

Modulating Density of Catalytic Sites in Multiple-Component Covalent Organic Frameworks for Electrocatalytic Carbon Dioxide Reduction

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ABSTRACT: It is generally assumed that the more metal atoms in covalent organic frameworks (COFs) contribute to higher activity towards electrocatalytic carbon dioxide reduction (CO₂RR), and hindered us to explore the correlation between the density of catalytic sites and catalytic performances. Herein, we have constructed quantitative density of catalytic sites in multiple-COFs for CO₂RR, in which the contents of phthalocyanines (H₂Pc) and Ni-phthalocyanines (NiPc) units were precisely controlled. With the molar ratio of 1/1 for H₂Pc and NiPc units in the COFs, the catalyst achieved highest selectivity with the CO Faradaic efficiency (FE_{CO}) of 95.37% and activity with the turnover frequency (TOF) of 4713.53 h⁻¹. In the multiple H₂Pc/NiPc-COFs, the electron-donating features of H₂Pc units provide electrons transport to NiPc centers and thus improved the binding ability of CO₂ and intermediates on the NiPc units. The theoretical calculation further confirmed that the H₂Pc units donated their electrons to NiPc units in the frameworks, and enhance the electron density of Ni sites and improved the binding ability with Lewis acidic CO₂ molecules, thereby boosting the CO₂RR performance. This study provides us a new insight into design highly active catalysts in electrocatalytic systems.

Introduction

The electrocatalytic carbon dioxide reduction (CO₂RR) into value-added fuels and chemical feedstock is a prospective technology to achieve carbon neutralization.^[1-5] However, there are still several problems in the electrocatalysis process for CO₂RR, including the large energy which activated the inert CO₂, the multielectron transfer processes and the low availability of

catalytic sites led the poor product selectivity and the competitive hydrogen evolution reaction (HER).^[6-14] The catalytic properties not only depend on the kinds of metal centres and their coordination environments, but also rely on the density of the metal atoms.^[15-18] However, how to precisely control the contents of metal sites is hardly achieved in the carbon-based catalysts.^[19-22]

Covalent organic frameworks (COFs), as one kind of porous organic materials, have attracted considerable attention for various applications such as gas/molecular absorption, photo- / electro-catalysis, lithium- batteries and energy storage systems.^[23-34] With using various functional units, COFs have been utilized in oxygen reduction reaction, oxygen evolution reaction (OER), hydrogen evolution reaction (HER), H₂O₂ synthesis.^[35-39] Among catalytic COFs, Metal-Por and Metal-Pc units have been widely employed to catalyze CO₂RR since 2015.^[40] Till now, the different metal species, topologies, linkers and linkages have been adopted to improve the activity and selectivity.^[41-46] However, modulating the metal density in COFs to tune the catalytic behaviour for CO₂RR is ignored, because it is generally assumed that the more metals results in better catalytic activity. Thus, the correlation between the density of metal atoms and catalytic performances have not been explored.

Herein, we demonstrated for the first time the multiple-component synthesis strategy to construct the controllable metal density in the skeletons of COFs. This strategy enables a quantitative correlation between the metal density and catalytic performance and shows us how to optimize the activity and selectivity. To exclude the effects of linkers and linkages in the COFs for CO₂RR, all the H₂Pc and NiPc are confused with benzene units directly, without any other building units. The alternate H₂Pc in the skeletons enables to promote the electron transfer to the NiPc units. The optimized COF showed high activity and selectivity, with TOFs of 4909.87 h⁻¹ at -1.0 V and FE_{CO}

of 95.37% at -0.8 V, respectively. This opens a way to guide the predesign electrocatalysts with higher atomic utilization efficiency.

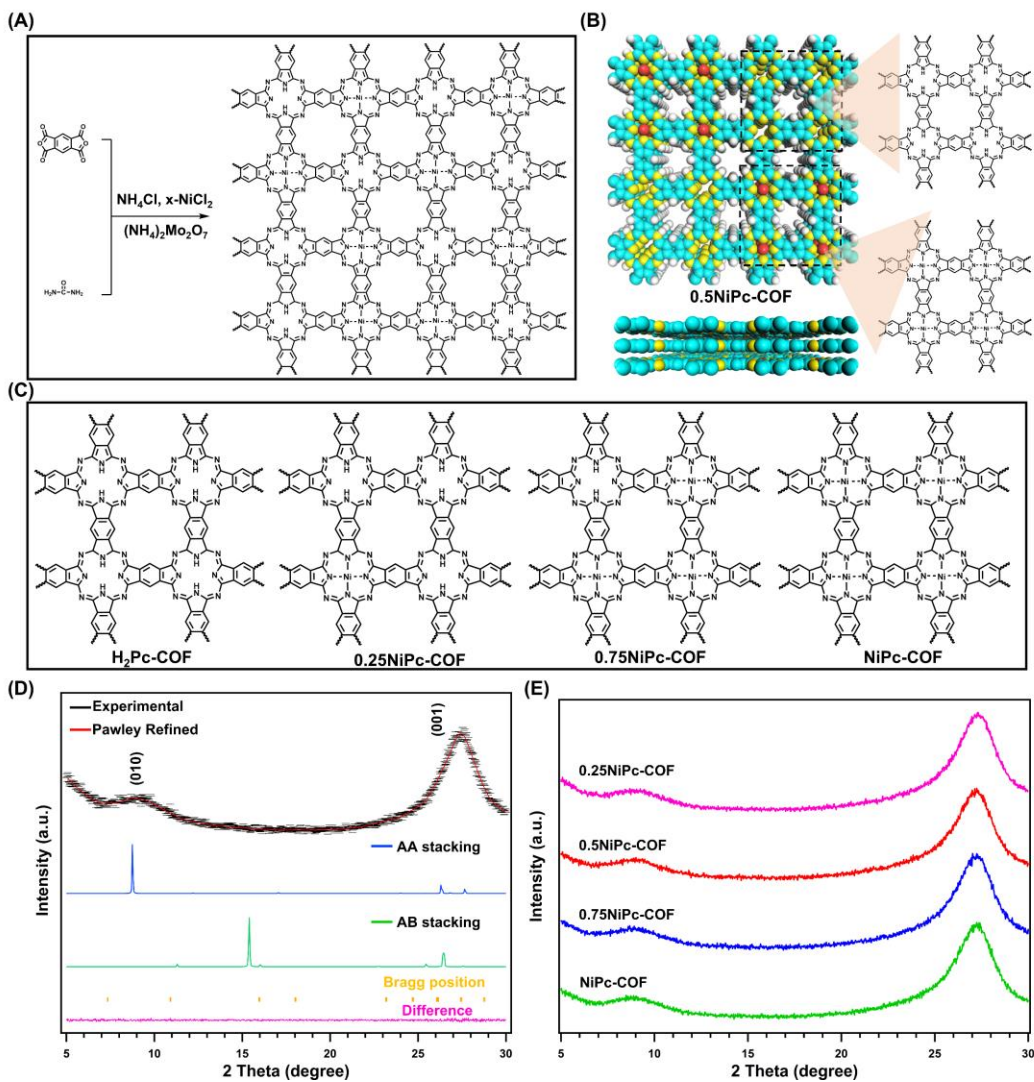


Figure 1. (A) Synthesis of XNiPc-COFs ($\text{H}_2\text{Pc-COF}$, 0.25NiPc-COF , 0.5NiPc-COF , 0.75NiPc-COF and NiPc-COF) from PADA, urea, NH_4Cl , NiCl_3 and $(\text{NH}_4)_2\text{Mo}_2\text{O}_7$, respectively, with (B) top view and side view of the slipped AA stacking structure (C-cyan, N-yellow, Ni-red) for 0.75NiPc-COF s. (C) The structures of $\text{H}_2\text{Pc-COF}$, 0.25NiPc-COF , 0.75NiPc-COF and NiPc-COF . (D) PXRD profiles of $\text{H}_2\text{Pc-COF}$ of the experimentally observed (black), Pawley refined

(red) and their difference (pink), simulated using the AA (blue) and staggered AB (green) stacking modes. (E) PXRD profiles of 0.25NiPc-COF (pink), 0.5NiPc-COF (red), 0.75NiPc-COF (blue) and NiPc-COF (green).

Experimental Methods

The metal-free COF (H₂Pc-COF) was synthesized from pyromellitic dianhydride (PMDA) (1.86 mmol), urea (40.10 mmol), (NH₄)₂M₂O₇ (1.17 mmol) and NH₄Cl (5.00 mmol) at 180 °C for 3 hours by solid-phase synthesis according to previous reports.^[47] To achieve different contents of Ni in the COFs, 0.21 mmol, 0.42 mmol, and 0.63 mmol NiCl₂ were added in the mixtures to yield XNiPc-COF, X = 0.25, 0.5, and 0.75 (0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF) respectively (Figure 1A-C). With additional of 0.84 mmol NiCl₂ in the mixture, the NiPc-COF was synthesized. The weight contents of Ni in the COFs revealed by ICP measurement were 3.03%, 5.75%, 7.81% and 9.57% for 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF, which were close to theoretical values (3.22%, 6.06%, 8.57% and 10.81%, respectively.).

Results and discussion

The structures of as-synthesized COFs were investigated using Fourier transform infrared (FT-IR) spectroscopy. It demonstrated that the obvious peak at about 1625 cm⁻¹ was assigned to C=N bonds for H₂Pc-COF.^[48] And the peaks at nearly 750 cm⁻¹ originated from the Ni-N bonds of XNiPc-COFs. These results suggested that the COFs have been successfully synthesized (Figure S1).

Powder X-ray diffraction (PXRD) measurement was conducted to confirm the crystallinities of these COFs. Exactly, the peaks at 8.36 and 27.5 ° were clearly observed, which were from (010)

and (001) facets of H₂Pc-COF (Figure 1D). Additionally, the theoretical structure was simulated by the Materials Studio package. The simulated PXRD patterns are in good agreement with experimental results (R_{wp} of 3.23% and R_p of 2.96%, respectively). Furthermore, the AA stacking models of H₂Pc-COF matched well with the experimental results and adopted the *P4M* space group, following the cell parameters of $a = 10.70 \text{ \AA}$, $b = 10.70 \text{ \AA}$, $c = 3.45 \text{ \AA}$, $\alpha = \beta = \gamma = 90^\circ$ (Figure S2 and Tables S1-S2). With immobilizing Ni atoms in the COFs (0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF), the obvious peaks from (010) and (001) were confirmed without position shift, confirming the crystal structures were as same as that of metal-free COFs (Figure 1E and Figures S3-S6).

The porous structures of five as-synthesized COFs were confirmed using nitrogen adsorption isotherm at 77 K. These H₂Pc-COF and XNiPc-COF demonstrated the curves with IV-types (Figure S7). Correspondingly, the Brunauer-Emmett-Teller (BET) surface areas for H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF were 24.32, 28.98, 28.11, 46.76 and 88.66 m² g⁻¹, with pore volumes of 0.11, 0.09, 0.08, 0.15 and 0.21 cm³ g⁻¹, respectively. The low permanent porosity could be attributed to the random displacement of the 2D layers to some extent.^[49]

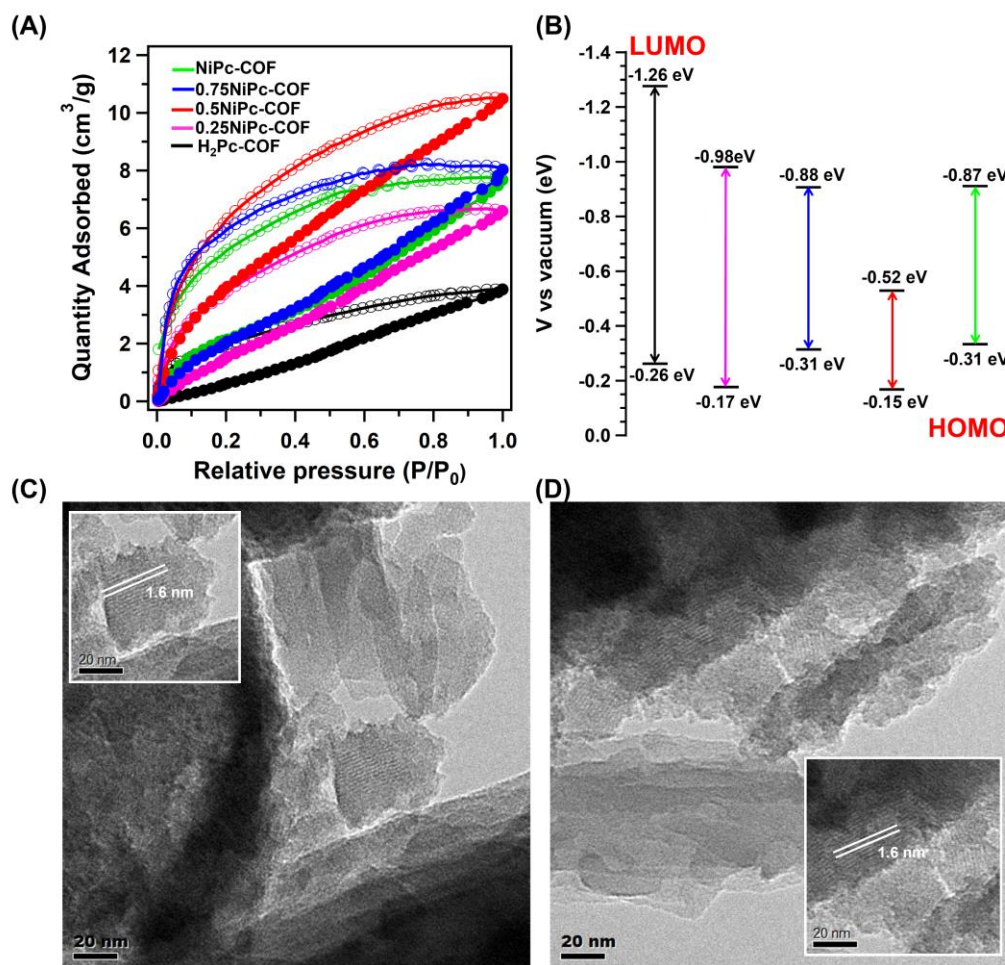


Figure 2. (A) The CO₂ adsorption curves at 298 K and (B) the energy gap (HOMO and LUMO) of H₂Pc-COF (black), 0.25NiPc-COF (pink), 0.5NiPc-COF (red), 0.75NiPc-COF (blue) and NiPc-COF (green). The TEM images of (C) H₂Pc-COF and (D) NiPc-COF.

Considering the important role of CO₂ adsorption ability for CO₂RR, the CO₂ sorption behaviour at 298 K was investigated. The CO₂ physical-sorption capacity depends on the surface areas and the chemical-sorption rely on the numbers of binding sites. With increasing the contents of Ni, the larger sorption loops suggested the higher binding ability of CO₂. The H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF had moderate CO₂ uptake capacities of 3.88, 6.60, 10.51, 8.31 and 8.04 cm³ g⁻¹ at 1.0 bar, respectively (Figures 2A). With introducing the Ni sites in

the COFs, the CO₂ uptake capacity were obviously improved, which is benefit for improving catalytic activity of CO₂RR.

To explore the properties of prepared XNiPc-COFs, the ultraviolet–visible (UV-Vis) spectroscopy was adopted (Figure S8). Specifically, the Tauc plots demonstrated that the H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF had band gaps of 0.39, 0.32, 0.18, 0.28 and 0.27 eV, respectively, indicating that 0.75NiPc-COF favoured the electron transfer in the process of electrocatalysis CO₂RR (Figure 2B).^[50] Then, the highest occupied molecular orbital (HOMO) and lowest unoccupied molecular orbital (LUMO) were identified using the Mott-Schottky method. The HOMO positions for H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF were -0.26, -0.17, -0.15, -0.31 and -0.31 eV, respectively, indicating that 0.75NiPc possessed better reduction performance than that of other COFs (Figure 2B and Figure S9). We also tested the conductivity of NiPc-COF and 0.5NiPc-COF by the four-probe method at 298 K. The electrical conductivity for NiPc-COF was determined to be $2.8 \times 10^{-5} \text{ S cm}^{-1}$, which is closed to the 0.5NiPc-COF ($2.5 \times 10^{-5} \text{ S cm}^{-1}$) (Figure S10).

The field-emission scanning electron microscopy (FE-SEM) demonstrated that H₂Pc-COF was in rodlike shape as the same the other four synthesized XNiPc-COFs (Figures S11-S15). In addition, the transmission electron microscopy (TEM) and high-resolution (HR-TEM) characterization of H₂Pc-COF exhibited the interlayer distance of 1.6 nm, confirming the crystal structure (Figure 2C). With the addition of nickel, NiPc-COF also exhibited a good crystallinity with the similar morphologies (Figure 2D and Figures S16-S18). These results suggested that there is no significant morphological difference with the introduction of nickel, further indicating that the structure is well preserved. Furthermore, the energy dispersive X-ray spectroscopy (EDX)

mapping images revealed that C, N and Ni are uniformly distributed over the COFs (Figures S19-S23).

Then, the chemical and thermal stability of XNiPc-COFs were studied. The XNiPc-COFs were immersed into harsh conditions, including HCl (1.0 M), NaOH (1.0 M) and KHCO₃ (0.5 M) for one week. The mass and crystallinity showed no obvious change from PXRD patterns, suggesting the well retained crystal structure (Figure S24). We have also used the thermogravimetric analysis (TGA) measurement to investigate the thermal stability, confirming that no obvious change up to 300 °C under N₂ of these XNiPc-COFs (Figure S25). Then, we also tested the thermal stability of these COFs under air atmosphere by the TGA measurement. The TGA curves demonstrated that the H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF left 29%, 27%, 25%, 22% and 20% after 550 °C under air (Figure S26).

Moreover, the hydrophobic characteristics of these COFs were conducted by the contact angle measurements with water (Figure S27). The contact angle of the H₂Pc-COF was 95.37±4.3°, while the 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF were 125.69±5.8°, 126.15±3.1°, 126.9±4.6° and 127.53±3.4°, respectively. The H₂Pc-COF demonstrated a lower water contact angle compared with that of NiPc-contained COFs, suggesting that water could be accessible to the surface, further facilitating proton transfer and hindering CO₂RR.^[51]

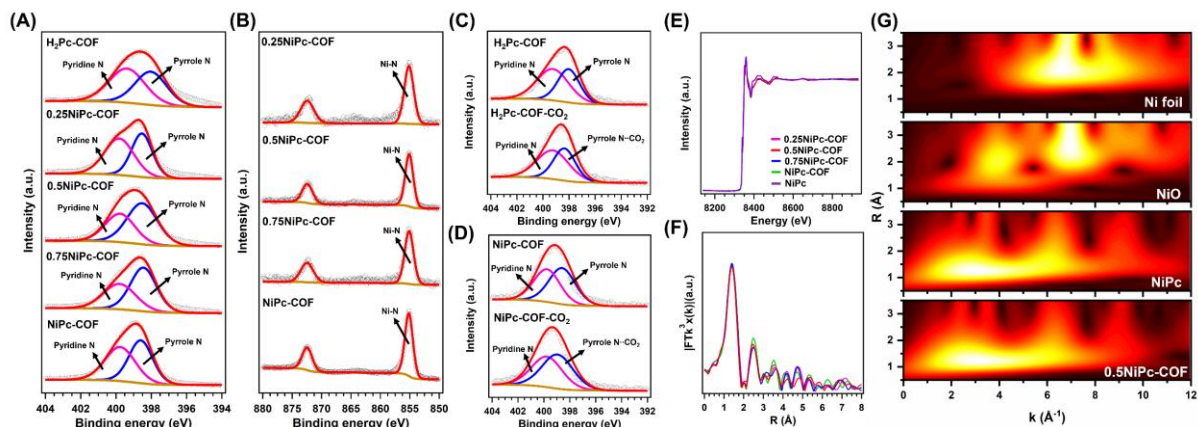


Figure 3. (A) The XPS spectra of N 1s and (B) Ni 2p for H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF. The XPS spectra of N 1s of (C) H₂Pc-COF and (D) NiPc-COF in-situ the CO₂ adsorption under the moderate pressure. (E) the XANES spectra and (F) the K-edge k^3 -weighted Fourier transform spectra from EXAFS for Ni of NiPc (purple curve), 0.25NiPc-COF (pink curve), 0.5NiPc-COF (red curve), 0.75NiPc-COF (blue curve) and NiPc-COF (green curve). (G) The WT-EXAFS of the Ni foil, NiPc and 0.5NiPc-COF.

X-ray photoelectron spectroscopy (XPS) was performed to explore the chemical structures and electron states of these COFs. Correspondingly, the high-resolution N 1s spectra of these H₂Pc-COF demonstrated peaks at 398.02 and 399.37 eV for pyrrole N and pyridine N, respectively. After the introduction of nickel, the N 1s spectra of 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF showed a positive shift compared with that of the H₂Pc-COF. Specifically, the pyridine N had a positive shift of about 0.3 eV, while the pyrrole N had no change, suggesting the successful coordination between pyridine N and nickel (Figure 3A). Additionally, the high-resolution Ni 2p spectra of 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF exhibited two distinct peaks at about 855.12 eV (2p_{3/2}) and 872.43 eV (2p_{1/2}), attributing to the Ni-N coordination (Figure 3B). Furthermore, the binding energy of Ni-N was slightly negative shift with 0.13 eV for

0.25NiPc-COF, 0.5NiPc-COF and 0.75NiPc-COF compared with that of NiPc-COF due to the H₂Pc units adjusted the local electron density of the active Ni-N sites, which benefits the adsorption of reactant molecules and transfers more electrons to the reaction.^[58]

To further study the CO₂ adsorption sites in these COFs, the in-situ XPS was measured under the ambient pressure with 0.1 mbar CO₂ atmosphere. Specifically, the high-resolution N 1s spectra of H₂Pc-COF demonstrated that the absence of shift for pyridine N, while the pyrrole N showed a positive shift with the value of 0.3 eV (Figure 3C). With immobilizing Ni atoms, the high-resolution N 1s spectra of NiPc-COF also exhibited the same trends with that of the H₂Pc-COF. In addition, the high-resolution N 1s spectra of other XNiPc-COFs also exhibited a positive shift of the pyridine N (Figure 3D and Figure S28). These results suggested that the pyrrole N as ‘CO₂ traps’ in the vacancies of the structure can adsorb CO₂ which can accelerate the electron supply and transport, further boosting the CO₂RR.^[52]

The X-ray absorption fine structure (XAFS) measurements were adopted to further confirm the electronic and atomic states of Ni in 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF. As shown in Figure 3E, the location of Ni absorption near-edge of 0.25NiPc-COF (pink curve), 0.5NiPc-COF (green curve), 0.75NiPc-COF (blue curve) and NiPc-COF (red curve) exhibited different structures from Ni foil and NiO (Figure S29), but they approached simple substances (NiPc) (purple curve). Furthermore, the peak positions of Ni in R-space about 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF were 4.9±0.7, 4.6±1.2, 4.8±1.5 and 4.7±1.4 Å, which was close to that of NiPc coordination (Figure 3F). In addition, the EXAFS fitting results demonstrated the existence of Ni-N₄ coordination for Ni atom in 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF (Figure S30 and Table S3). Additionally, to further confirm the combination of Ni, the wavelet transform (WT) of the Ni L³-edge EXAFS oscillations

was conducted (Figure 3G). 0.5NiPc-COF and NiPc exhibited an intensity maximum at 2.3 \AA^{-1} (Ni-N), while only a predominate intensity maximum at higher k-space (6.2 \AA^{-1}) for Ni foil and 6.5 \AA^{-1} for NiO, suggesting the absence of the Ni-Ni or Ni-O metallic coordination, further proving that no metal nanoparticles are present in four COFs.

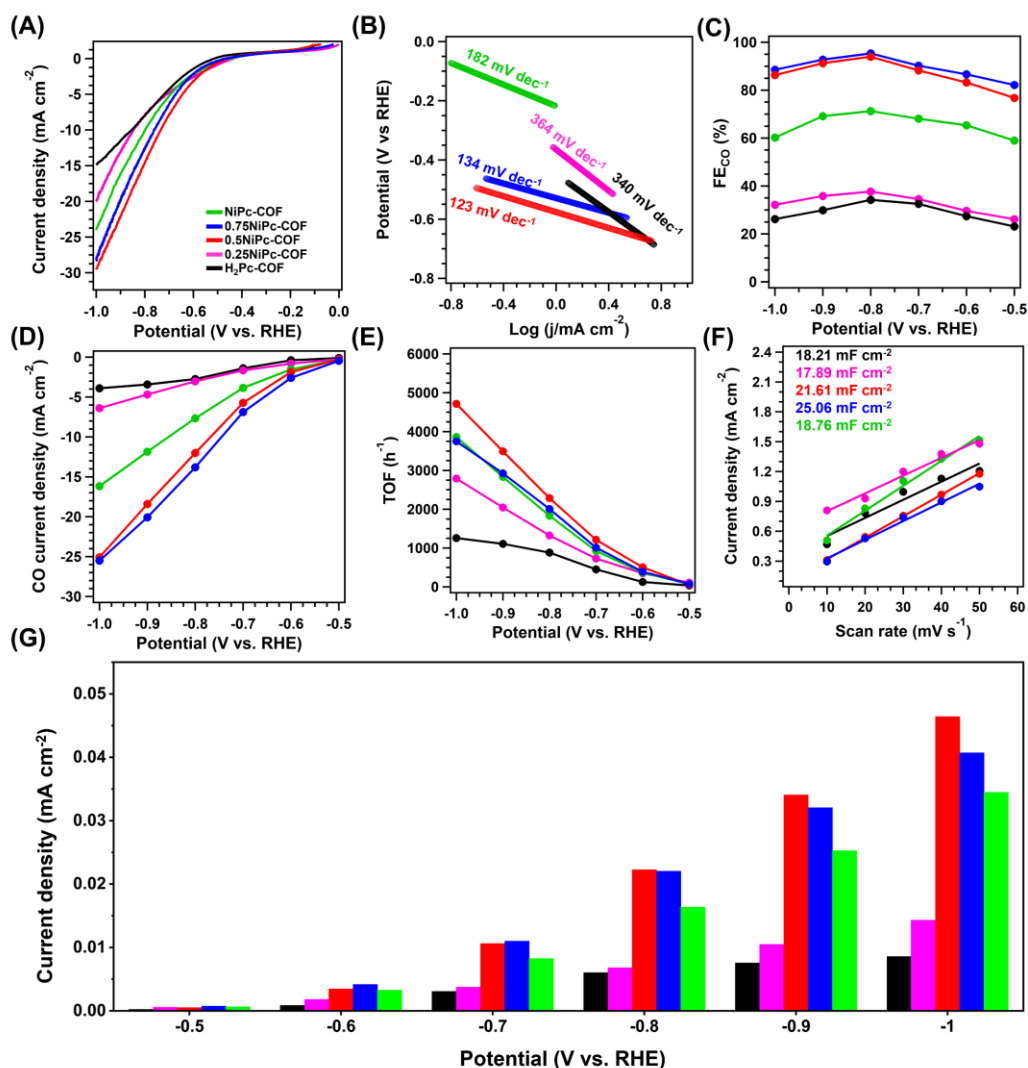


Figure 4. (A) LSV curves, (B) Tafel slopes, (C) CO faradaic efficiency, (D) the partial CO current density, (E) the corresponding TOF values, (F) the ECSA slopes and (G) the current density normalized by ECSA for H₂Pc-COF (black curve), 0.25NiPc-COF (pink curve),

0.5NiPc-COF (red curve), 0.75NiPc-COF (blue curve), NiPc-COF (green curve) from -0.5 to -1.0 V in 0.5 M KHCO_3 under CO_2 atmosphere.

To further investigate the CO_2RR performance of the XNiPc-COFs, using a H-cell separated by a Nafion-117 membrane in the 0.5 M KHCO_3 solution with a three-electrode configuration under the saturated CO_2 . The linear sweep voltammetry curves (LSV) demonstrated that these XNiPc-COFs had a small onset potential (Figure 4A). The kinetic behaviours for synthesized COFs were revealed by Tafel slopes (Figure 4B). And the Tafel slope of the 0.5NiPc-COF, 0.75NiPc-COF were 134 and 123 mV dec^{-1} , which much lower than that of $\text{H}_2\text{Pc-COF}$, 0.25NiPc-COF and NiPc-COF with values of 340 , 364 and 182 mV dec^{-1} , respectively, indicating that the formation of $^*\text{COOH}$ from the adsorbed CO_2 via a proton-coupled electron transfer process is the rate determining step (RDS) for the 0.5NiPc-COF and 0.75NiPc-COF.^[53] However, the much higher Tafel slopes of $\text{H}_2\text{Pc-COF}$, 0.25NiPc-COF and NiPc-COF suggested the slow kinetic process.

To explore the selectivity of COFs for CO_2RR , the gas products at different applied potentials from -0.5 to -1.0 V for 300 s were analysed (Figure S31). It showed CO and H_2 were observed and no liquid product was produced in the CO_2RR process. The Faradaic efficiencies of CO (FE_{CO}) for the $\text{H}_2\text{Pc-COF}$ were 23.12% , 27.41% , 32.56% , 34.21% , 29.88% and 26.18% from -0.5 to -1.0 V, respectively (Figure 4C, black curve). The FE_{CO} of the conductive carbon were 2.22% , 3.12% , 6.13% , 9.25% , 3.34% and 4.02% at -0.5 , -0.6 , -0.7 , -0.8 , -0.9 and -1.0 V, respectively, suggesting its little contribution of CO_2RR performance (Figure S32). In addition, we calculated the partial CO current density (j_{CO}) to evaluate the activity. The $\text{H}_2\text{Pc-COF}$ demonstrated the maximum j_{CO} of 3.91 mA cm^{-2} at -1.0 V. The selectivity was highly improved with the increasement of Ni-N sites of the structure, suggesting that H_2Pc units were not the active origin for CO_2RR (Table S4).

Although 0.25NiPc-COF has delivered more positive E_0 , the selectivity and activity for CO products are still limited. And the maximum FE_{CO} was 37.66% at -0.8 V with a highest J_{CO} of 6.34 mA cm^{-2} at -1.0 V, suggesting the more H_2Pc units in the frameworks caused H_2 as the major products. When the ratio between the H_2Pc and NiPc were 1/1 in 0.5NiPc-COF, the activity and selectivity for producing CO showed obvious enhanced (the maximum FE_{CO} was 92.52% at -0.8 V with a highest J_{CO} of 25.08 mA cm^{-2} at -1.0 V) (Figure 4D, red curve). The 0.75NiPc-COF had higher selectivity than that of other XNiPc-COF in the same potential range, with FE_{CO} of 82.19%, 86.62%, 90.22%, 95.37%, 92.77%, and 88.54%, respectively (Figure 4C, blue curve). The highest j_{CO} was 25.50 mA cm^{-2} at -1.0 V, which was close to that of the 0.5NiPc-COFs, (Figure 4D, blue curve). The FE_{CO} for the NiPc-COF were 65.97%, 69.32%, 73.12%, 75.68%, 73.17%, and 67.21% at the same potentials (Figure 4C, red curve), and the highest j_{CO} was 16.17 mA cm^{-2} at a potential of -1.0 V, which was much lower than that of 0.5NiPc-COF and 0.75NiPc-COF (Figure 4D, green curve). These results further indicated that the alternate structure of H_2Pc and NiPc can improve the CO selectivity due to the H_2Pc units adjusted the local electron density of the active Ni-N sites, further improving the CO_2RR . We also tested the CO_2RR performance of 0.5NiPc-COF in the acidic aqueous solution under CO_2 -atmosphere. Specifically, the FE_{CO} of the 0.5NiPc-COF were 19.43%, 21.02%, 22.11%, 20.43%, 19.31% and 17.52% at -0.3 , -0.4 , -0.5 , -0.6 , -0.7 and -0.8 V, respectively, indicating the lower activity in acidic solutions (Figure S33). The turnover frequencies (TOFs) of these XNiPc-COFs were calculated in the potential range of -0.5 to -1.0 V based on the loading of Ni-N sites in the XNiPc-COFs and N sites in the H_2Pc -COF as the catalytic origin. The calculated TOF values for the H_2Pc -COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF were 1258.64, 2790.41, 4713.53, 3749.88 and 3860.12 h^{-1} at -1.0 V, respectively (Figure 4E). Thus, the alternate structure of H_2Pc and NiPc with the ratio of 1/1 for

the 0.5NiPc-COF, which possessed high utilization efficiency of Ni-N sites, facilitated both its activity and selectivity.

Giving that the different catalytic behaviours of the XNiPc-COFs, the electrochemical active surface areas (ECSAs) were measured. Specifically, the electrochemical double layer capacitances (C_{dl}) were obtained using cyclic voltammogram (CV) plots (Figure S34). The C_{dl} values for the H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF were 18.21, 17.89, 21.61, 25.06, and 18.76 mF cm⁻², respectively (Figure 4F). And the ECSA were then calculated by $ECSA = C_{dl}/C_s$, in which the C_s is 0.04 mF cm⁻².^[54-57] Therefore, the evaluated ECSA for H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF were 455.25, 447.25, 540.25, 626.50 and 469. To investigate the different surface catalytic activity, we have normalized the J_{co} by ECSA, and the corresponding current density for the catalytic COFs at different potentials were in the order of 0.5NiPc-COF > 0.75NiPc-COF > NiPc-COF > 0.25NiPc-COF > H₂Pc-COF at different potentials, which further confirming the highest activity for 0.5NiPc-COF (Figure 4G).^[58]

Furthermore, electrochemical impedance spectroscopy (EIS) was investigated. Nyquist plots and corresponding parameters showed that with the increasement of Ni-N sites, the 0.25NiPc-COF (61.18 Ω), 0.5NiPc-COF (33.27 Ω), 0.75NiPc-COF (59.71 Ω) and NiPc-COF (58.31 Ω). Specifically, they exhibited lower charge transfer resistance than that of the H₂Pc-COF (101.33 Ω), indicating that they were capable of faster electron transfer (Figure S35 and Table S5). In addition, the H₂Pc-COF showed the lowest diffusive resistance (3.32 Ω) compared with 0.25NiPc-COF (3.82 Ω), 0.5NiPc-COF (6.27 Ω), 0.75NiPc-COF (6.12 Ω) and NiPc-COF (7.02 Ω), revealing that the H₂Pc could ensure faster electron transfer from the electrodes to CO₂.

The long-term stability of the 0.5NiPc-COF was investigated at -0.8 V in CO_2 -saturated KHCO_3 for 24 h (Figure S36). Specifically, a slight attenuation of FE_{CO} from 95.37% to 90.68% with a steady current density of 13.6 mA cm^{-2} , indicating the excellent long-term stability. After the long-term stability measurement, the PXRD patterns and FT IR and XPS spectra were obtained to confirm the structure. Specifically, the PXRD patterns revealed no obvious change of these peaks, indicating the maintenance of the structure. (Figure S37). Meanwhile, after the durability test, the Ni 2p and N 1s peaks observed, corresponding to are in good agreement with those found for fresh 0.5NiPc-COF (Figures S38-S39). Furthermore, the FT IR spectra disclose the lack of any significant band change between fresh 0.5NiPc-COF and used catalyst (Figure S40). These results demonstrated their remarkable durability of these XNiPc-COFs in electrocatalysis CO_2RR process.

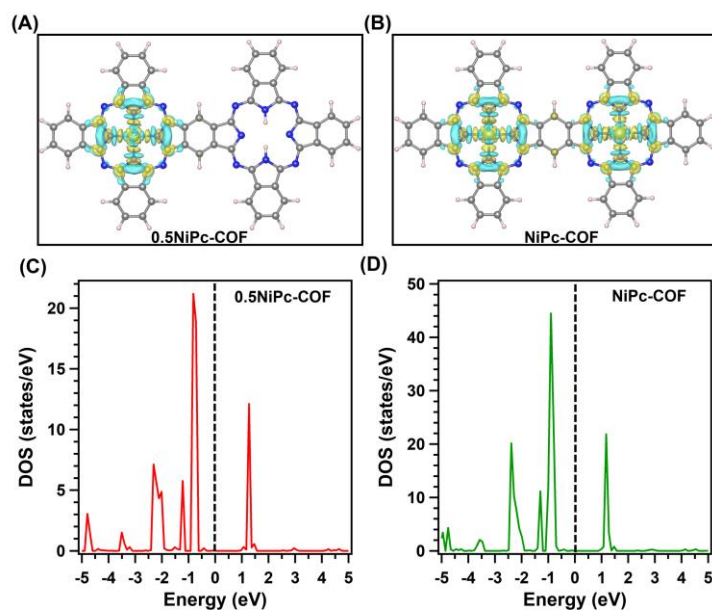


Figure 5. The electron density difference of (A) 0.5NiPc-COF and (B) NiPc-COF (The yellow and blue areas represent a gain and loss of electrons, respectively.). The projected density of states (pDOS) plots of (C) 0.5NiPc-COF and (D) NiPc-COF.

Density Functional Theory (DFT) calculations were adopted to further explain why 0.5NiPc-COF provides a better CO₂RR performance than NiPc-COF. Based on the experimental results, Ni-N was considered to be the active site for the CO₂RR, and 0.5NiPc-COF with a ratio of H₂Pc and NiPc of 1:1 and NiPc-COF structure models were constructed. The charge density difference of 0.5NiPc-COF clearly revealed that the Ni-N sites lost electrons (0.949 e⁻) when the intermediates adsorption, while the NiPc-COF lost 1.062 e⁻ in this process, indicating the alternate structure of H₂Pc and NiPc could weaken the CO₂RR energy barrier (Figure 5A and 5B). The H₂Pc units donated their electrons to NiPc units in the frameworks, and enhance the electron density of Ni sites with greater nucleophilicity and a stronger bond with Lewis acidic CO₂ molecules. In addition, the projected density of states (pDOS) plots were obtained for 0.5NiPc-COF and NiPc-COF (Figure 5C and 5D). The d band of the Ni-N site of 0.5NiPc-COF accommodates more electrons than that of NiPc-COF, further confirming that the electrons from H₂Pc are successfully transferred to NiPc. Furthermore, we used DFT to calculate the electron static potential (ESP) diagram of NiPc-COF and 0.5NiPc-COF. Specifically, ESP diagrams of NiPc-COF and 0.5NiPc-COF model revealed that the H₂Pc units can regulate the charge distribution. The electron-rich region (negative potential) of 0.5NiPc-COF is mainly distributed around the whole skeleton, while the pyridinic nitrogen of the NiPc-COF is more electron rich, indicating the formation of an internal electric field further promoting the CO₂RR of 0.5NiPc-COF (Figure S41). Additionally, the multiple components of H₂Pc and NiPc in the 0.5NiPc-COF can modulate the dipole moment. Specifically, the 0.5NiPc-COF (0.12 Debye) inhibits charge recombination and promotes charge transfer, resulting in a higher local charge redistribution than H₂Pc-COF (0.0006 Debye) and NiPc-COF (0 Debye) (Figure S42).^[59-61]

Conclusion

In this study, a skeleton engineering strategy based on the multiple-component synthesis to construct the controllable metal density in the skeletons was first reported, which build the quantitative correlation between the metal density and catalytic performance. By establishing COFs with of vacancies and Ni-N sites, the 0.5NiPc-COF catalyzed CO₂RR with high selectivity and activity (FE_{CO} of 95.37% at -0.8 V and TOF of 4713.53 h⁻¹ at -1.0 V). The design and modulation density of catalytic sites in COFs proposed not only new insights into electrocatalysis or energy conversion, but also guidance of constructing COFs with alternate structures of vacancies and metal sites for achieving high catalytic performance.

ASSOCIATED CONTENT

Supporting Information. Descriptions of characterization equipment and instrument.

Synthesis methods of H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF. Electrocatalysis CO₂RR performance tests. Mott-Schottky curve test. The calculation methods of FE_{CO} and TOF values. XAFS measurements and data processing. The Staggered-AB model of H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF. Figures of FT IR, N₂ adsorption–desorption isotherms, UV-Vis spectra, Tauc plots, SEM, TEM, EDX-mapping images, chemical stability test, thermogravimetric curves, water contact angles, the XPS spectra of N 1s in-situ the CO₂ adsorption under the moderate pressure for H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF. The XANES spectra for Ni of Ni foil NiO. EXAFS fitting curves for Ni in NiPc, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF. The LSV curves in N₂ atmosphere, the CV curves from 10 mV s⁻¹ to 50 mV s⁻¹, Nyquist plots of H₂Pc-COF, 0.25NiPc-COF, 0.5NiPc-COF, 0.75NiPc-COF and NiPc-COF. Chronoamperometry test for CO₂RR, PXRD, FT IR spectra, XPS of N 1s, Ni 2p for 0.5NiPc-COF. The dipole moment calculation of H₂Pc-COF 0.5NiPc-COF and NiPc-COF.

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Author Contributions

Q. Xu. conceived the idea and designed the experiments. M. Liu. performed the experiments. Y. Fu. contributed to the theory calculation part. S. Yang performed the EXAFS experiments. M. Liu, Q. Xu. and G. Zeng. wrote and revised the manuscript. All the authors contributed to the data interpretation, discussion, and manuscript revision. All authors have given approval to the final version of the manuscript. & M. Liu. X. Zhao and S. Yang. contributed equally to this work.

NOTES

The authors declare no competing financial interest.

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