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# Fast, multicolor photodetection with graphene-contacted p-GaSe/n-InSe van der Waals heterostructures

Faguang Yan<sup>1</sup>, Lixia Zhao<sup>2,3</sup>, Amalia Patanè<sup>4</sup>, PingAn Hu<sup>5</sup>, Xia Wei<sup>1</sup>, Wengang Luo<sup>1</sup>, Dong Zhang<sup>1</sup>,

Quanshan Lv<sup>1</sup>, Qi Feng<sup>1</sup>, Chao Shen<sup>1,3</sup>, Kai Chang<sup>1,3</sup>, Laurence Eaves<sup>4</sup> and Kaiyou Wang<sup>1,3</sup>

<sup>1</sup> State Key Laboratory of Superlattices and Microstructures, Institute of Semiconductors, Chinese Academy of Sciences, Beijing 100083, China

<sup>2</sup> State Key Laboratory of Solid-State Lighting, Institute of Semiconductors, Chinese Academy of Sciences, Beijing 10083, China

<sup>3</sup> College of Materials Science and Opto-Electronic Technology, University of Chinese Academy of Science, Beijing 100049, China

<sup>4</sup> School of Physics and Astronomy, University of Nottingham, Nottingham NG7 2RD, UK.

<sup>5</sup> Key Lab of Microsystem and Microstructure, Harbin Institute of Technology, Ministry of Education, Harbin, 150080, China

**E-mail:** kywang@semi.ac.cn

**Keywords:** Multicolor, gallium selenide, indium selenide, van der Waals heterostructure, built-in potential

## Abstract

The integration of different two-dimensional materials within a multilayer van der Waals (vdW) heterostructure offers a promising technology for high performance opto-electronic devices such as photodetectors and light sources. Here we report on the fabrication and electronic properties of vdW heterojunction diodes composed of the direct band gap layered semiconductors InSe and GaSe and transparent monolayer graphene electrodes. We show that the type II band alignment between the two layered materials and their distinctive spectral response, combined with the short channel length and low electrical resistance of graphene electrodes, enable efficient generation and extraction of photoexcited carriers from the heterostructure even when no external voltage is applied. Our devices are fast ( $\sim 1 \mu\text{s}$ ), self-driven photodetectors with multicolor photoresponse ranging from the ultraviolet to the near-infrared and offer new routes to miniaturized optoelectronics beyond present semiconductor materials and technologies.

## Introduction

Multicolor photodetectors covering the ultraviolet (UV), visible and infrared (IR) spectral ranges have potential for a wide range of applications, such as optical communication [1, 2], imaging [3], environmental monitoring [4] and astronomical observations [5]. Furthermore, robust and miniaturized self-driven devices, which require no electrical power source, are of particular interest for applications in extreme conditions. Self-driven multicolor photodetectors using semiconductor heterojunctions, such as  $\text{MoS}_2/\text{Si}$  [6],  $\text{CuO}/\text{Si}$  [7],  $\text{Bi}_2\text{Se}_3/\text{Si}$  [8] and  $\text{MoS}_2/\text{GaAs}$  heterojunctions [9], have attracted considerable attention recently. Photodetectors based on van der Waals (vdW) heterostructures have also been demonstrated [10, 11]. However, self-driven “multicolor” photodetectors that require no external power

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4 source are more difficult to realize. VdW heterostructures, which can be assembled by stacking different  
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6 two-dimensional (2D) semiconductors with different bandgaps, can combine and exploit the properties of  
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8 the component materials within a single device. Such structures are therefore candidates for  
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10 multifunctional optoelectronic systems with superior performance. In contrast to gapless graphene [12],  
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12 GaSe [13] and InSe [14] and monolayers of the transition metal dichalcogenides (TMDCs) [15,16], are  
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14 direct band gap semiconductors. High performance photodetectors based on few-layered *p*-type GaSe or  
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16 *n*-type InSe have been reported previously [17-19]. In addition to their response to visible light, the  
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18 photoresponse of 2D GaSe photodetectors [13] can extend into the ultraviolet (UV) region, while InSe  
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20 nanosheets show strong near-infrared (NIR) photoluminescence (PL) emission and photoresponsivity [20].  
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22 These results suggest that a heterojunction based on 2D *p*-GaSe and *n*-InSe could be used for  
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24 photodetection over a still broader spectral range.  
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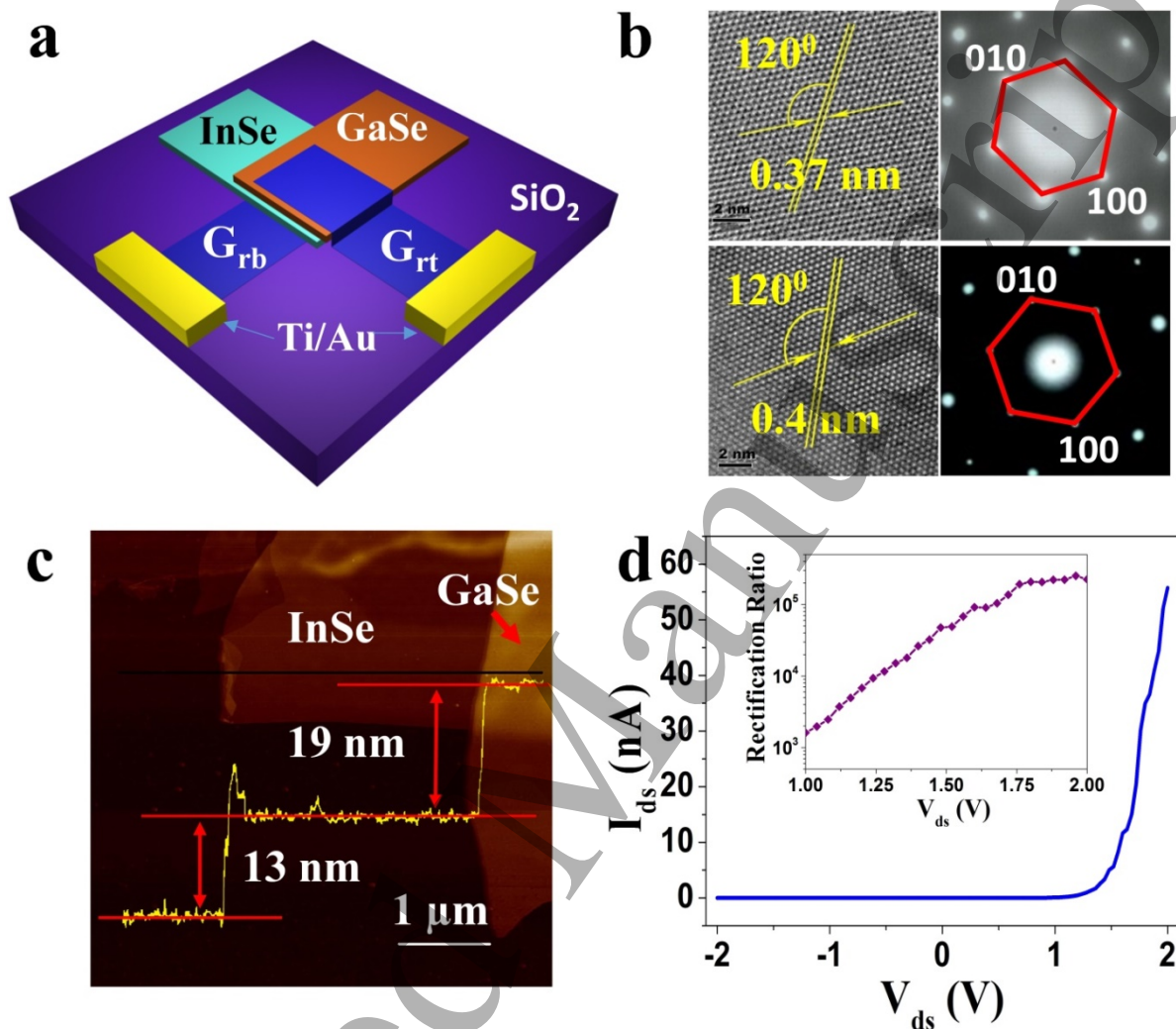
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34 Although vdW heterostructures with metal electrodes have been studied extensively and demonstrate  
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36 interesting optoelectronic and electronic properties [21, 22], their response time ranges from milliseconds  
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38 [23] to seconds [24]. To fabricate faster, higher-performance devices, it is essential that the optically  
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40 active layers have good interfaces and Ohmic contacts. In contrast to metal-contacted photodetectors, the  
41  
42 near perfect optical transparency and the unique electronic properties of graphene make it an ideal  
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44 electrode for multilayer, “vertical” vdW heterostructures as it can act as a short, atomically thin charge  
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46 extraction channel with a large active area, thus enabling both fast and efficient photodetection [19]. An  
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48 effective method to create faster vdW heterostructure photodetectors is therefore to sandwich the  
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50 heterojunction between two layers of graphene, which act as electrodes. These heterostructures can be  
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52 fabricated with clean interfaces free from dangling bonds, with low defect density and without the  
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4 Fermi-level pinning that often occurs when metal contacts are directly deposited onto a semiconductor  
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6 surface.  
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9 Here we report on graphene contacted  $p$ -GaSe/ $n$ -InSe heterojunctions. A typical device structure is  
10 shown schematically in figure 1(a). A GaSe layer is placed directly on top of the InSe sheet. This  
11 sequence of layers ensures that photons of energy  $h\nu < 2$  eV are transmitted through the wide band gap  
12 energy GaSe ( $E_g = 2.05$  eV at 300K) and can excite electron-hole pairs in the smaller band gap InSe ( $E_g =$   
13  $1.26$  eV at 300K). This large area ( $\sim 50 \mu\text{m}^2$ )  $p$ -GaSe/ $n$ -InSe heterojunction exhibits a strong self-driven  
14 photoresponse ranging from the UV to NIR due to the built-in potential in the heterojunction, the type-II  
15 band alignment between the two layered crystals [25] and their distinctive band gap energies. Furthermore,  
16 using graphene rather than metals as electrodes enable a response time as short as  $1.85 \mu\text{s}$ , *i.e.*  
17 significantly faster than that reported recently for van der Waals diode-like photodetectors [10, 22, 23],  
18 and 3 to 5 orders of magnitude faster than previously reported for photodetectors based on GaSe [13, 26,  
19 27] or InSe alone [19, 28], which usually have a slow response due to the presence of carrier traps in the  
20 relatively long active region of the detector [1, 29, 30].  
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### 43 **Results and discussion**

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45 Figure 1(b) shows high-resolution transmission electron microscopy images and electron diffraction  
46 patterns of the  $\beta$ -GaSe and  $\beta$ -InSe layers. These have high crystalline quality and in-plane hexagonal  
47 symmetry. Their crystal structure consists of Se-M-M-Se (M represents Ga- and In- atoms) layers, as  
48 shown in the Supplementary Information figure S1. The measured in-plane lattice constants of GaSe and  
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56 InSe are  $a = 0.37$  and  $0.4$  nm, respectively. The separations of two neighboring tetralayers of GaSe and  
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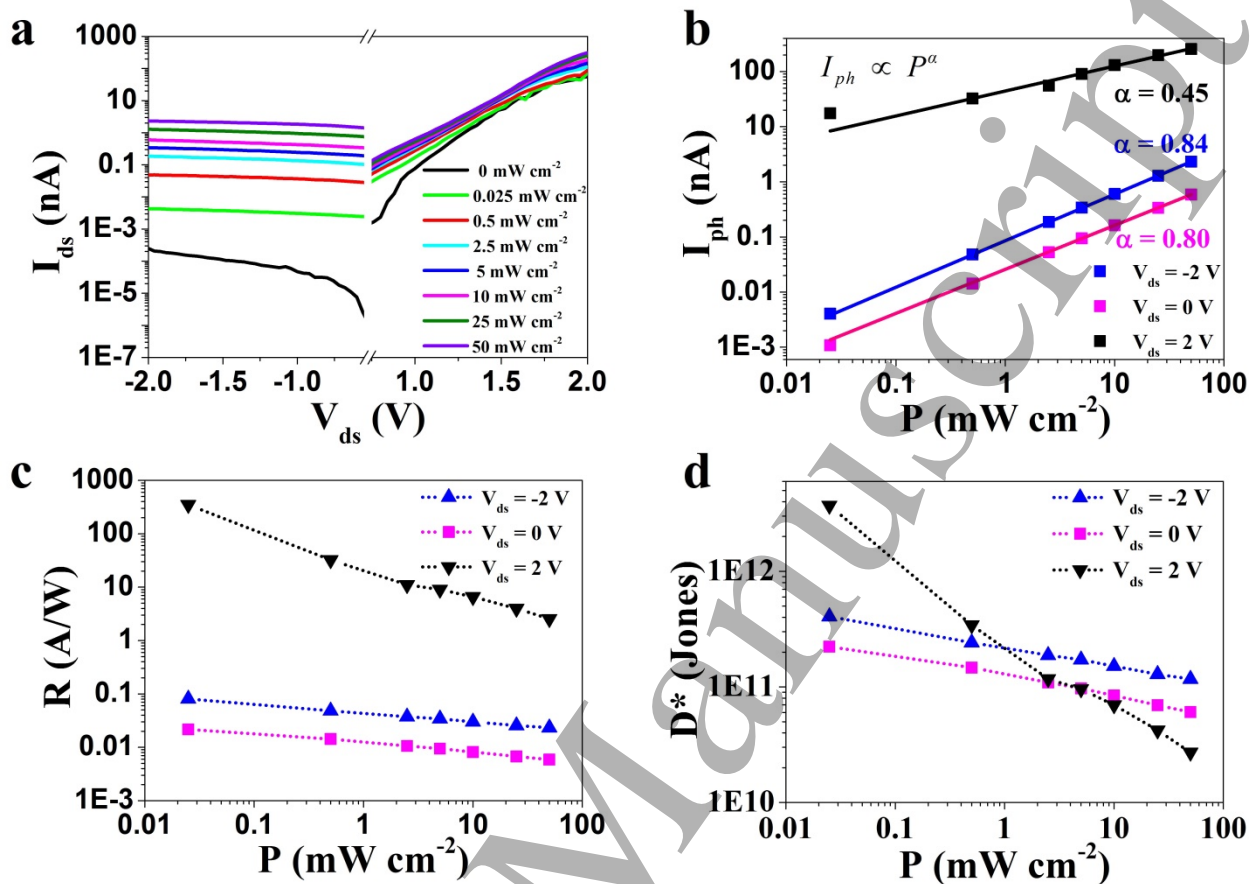
**Figure 1.** Schematic diagram and current-voltage  $I$ - $V$  characteristic of the  $p$ -GaSe/ $n$ -InSe heterojunction diode. (a) Schematic diagram of the  $p$ -GaSe/ $n$ -InSe heterojunction diode. (b) High-resolution TEM image of the GaSe (top left) and InSe (bottom left), respectively. Images on the right show the corresponding electron-beam diffraction patterns of GaSe and InSe. (c) AFM image of the device. The inset shows the thickness of the different layers. (d) The  $I$ - $V$  characteristic of the  $p$ -GaSe/ $n$ -InSe heterojunction diode at room temperature. The inset shows the rectification ratio as a function of the source-drain voltage  $V_{ds}$ .

InSe are  $d = 0.9$  and  $0.84$  nm, respectively. For the fabrication of the heterostructure, graphene microstamps were first transferred onto a fused silica substrate to form one electrode. The InSe flake was

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4 mechanically exfoliated using adhesive tape from bulk single crystals onto a stamp and then transferred  
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7 onto the graphene electrode. Using the same method, the GaSe sheet was transferred on top of the InSe  
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10 sheet. Finally, a second graphene microstamp was transferred onto the GaSe sheet to form the top  
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12 electrode (see also the Supplementary Information). Room temperature measurements of the electrical  
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14 properties of the heterojunction diode reveal strong rectification in the current-voltage ( $I$ - $V$ )  
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16 characteristics, with a larger current passing when the  $p$ -type GaSe is positively biased relative to  $n$ -type  
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18 InSe (figure 1(d)). The rectification ratio, defined as the ratio of the forward/reverse current, reaches  $\sim 10^5$   
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20 at  $V_{ds} = +2/-2$  V (figure 1(d), inset), demonstrating that a good  $p$ - $n$  diode is formed within the atomically  
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22 thin GaSe/InSe heterojunction.  
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28 Figure 2 shows the dependence of the  $I$ - $V$  characteristics on light intensity  $P$  ranging from 0 to 50 mW  
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30  $\text{cm}^{-2}$ . The source-drain current  $I_{ds}$  increases with increasing  $P$  (figure 2(a)) and the photocurrent  $I_{ph}$   
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32 exhibits sublinear behavior, *i.e.*  $I_{ph} \propto P^\alpha$ , where  $\alpha = 0.84, 0.80$  and  $0.45$  at source-drain voltages of  $V_{ds}$   
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34  $= -2, 0$  and  $2$  V, respectively (figure 2(b)). A similar sublinear response has also been reported for other  
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36 nanomaterials, such as ZnO [31] and GaN nanowires [32], WSe<sub>2</sub>/Graphene [30] and MoS<sub>2</sub>/WS<sub>2</sub>  
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38 heterostructures [33]. This response suggests a decrease of the recombination time of carriers with  
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40 increasing  $P$  due Auger recombination processes. We also note that in forward bias, due to the high  
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42 injection of majority carriers across the junction, Auger recombination could be enhanced thus leading to  
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44 a different power dependence of the photocurrent.  
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52 Figures 2(c) and (d) plot the photoresponsivity ( $R = I_{ph}/PS$ ) and detectivity ( $D^* = RS^{1/2}/(2eI_{dark})^{1/2}$ ) of the  
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54 heterojunction at different applied voltages as a function of incident light intensity. Here  $S$  is the in-plane  
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56 area ( $50 \mu\text{m}^2$ ) of the device,  $e$  is the electron charge and  $I_{dark}$  is the dark current. Both  $R$  and  $D^*$  increase  
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**Figure 2.** Power-dependent optoelectronic characterization at different applied voltages  $V_{ds}$ . (a) Typical  $I_{ds}$  curves of the p-GaSe/n-InSe heterojunction diode with illumination at various excitation intensities ( $P = 0, 0.025, 0.5, 2.5, 5, 10, 25, 50 \text{ mW cm}^{-2}$ ) and wavelength  $\lambda = 410 \text{ nm}$  at room temperature. (b) Photocurrent as a function of the illumination intensity at different  $V_{ds}$  (forward bias  $V_{ds} = 2 \text{ V}$ , zero bias  $V_{ds} = 0 \text{ V}$  and reverse bias  $V_{ds} = -2 \text{ V}$ ). The solid lines are fits to the data. (c,d) Photoresponsivity  $R$  (c) and detectivity  $D^*$  (d) of the heterojunction diode as a function of the illumination intensity  $P$  at different  $V_{ds}$  ( $V_{ds} = 2, 0, -2 \text{ V}$ ). The spot size of the laser beam is about  $0.2 \text{ mm}^2$ , which is much larger than the device size.

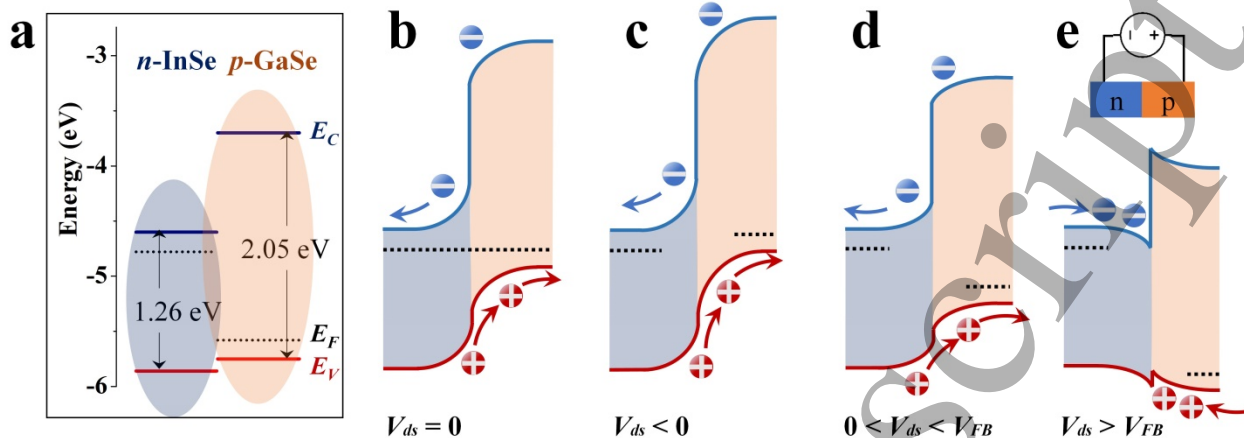
with decreasing light intensity, consistent with the sublinear behavior of the photocurrent. The decrease of  $R$  with increasing  $P$  is common to many photodetectors. It suggests a decrease of the recombination time



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4 of carriers with increasing  $P$  due Auger recombination processes; further, an increasing  $P$  can also  
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6 increase the carrier transit time due to increased carrier scattering.  
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10 The response of a photodetector is determined by a combination of several processes, including the  
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12 excitation, recombination, and diffusion of carriers. Due to the built-in potential of the heterojunction and  
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14 the type II band alignment (figure 3(a)), at zero bias the photocreated electrons and holes are swept in  
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16 opposite directions across the junction into the graphene electrodes (figure 3(b)). We estimate that the  
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18 built-in potential of the heterostructure is about 0.6 V (see Supplementary Information, figure S2). Thus  
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20 our devices can operate at zero bias with a photoresponsivity of up to  $R = 21 \text{ mA W}^{-1}$  with corresponding  
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22 detectivity  $D^* = 2.2 \times 10^{12}$  Jones at  $\lambda = 410 \text{ nm}$ . The systematic decrease of  $R$  and  $D^*$  with increasing  $P$   
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24 can arise from stronger Coulomb interactions between the photogenerated carriers and enhanced  
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26 radiative/non-radiative recombination. A reverse bias voltage increases  $I_{ph}$  and  $R$  due to the increased  
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28 electric field in the junction, which decreases the carrier transit time, resulting in reduced carrier  
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30 recombination (figure 3(c)). We also note that  $I_{ph}$  and  $R$  are strongly enhanced in applied forward biases  
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32 beyond the open circuit voltage  $V_{oc}$  (e.g.  $I_{ds} = 0$ ) and at high  $V$ . In this regime, due to the high injection of  
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34 majority carriers across the junction, the influence of carrier traps is weaker and a larger number of  
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36 photogenerated carriers are effectively extracted across the thin layers into the graphene electrodes  
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38 (figures 3(d) and (e)).  
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50 In our devices, a photoresponsivity of up to  $R = 350 \text{ A W}^{-1}$  is obtained at  $V_{ds} = 2 \text{ V}$  with an illumination  
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52 intensity  $P = 0.025 \text{ mW cm}^{-2}$  and  $\lambda = 410 \text{ nm}$ . This value of  $R$  is 2 to 3 orders of magnitude larger than for  
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54 heterojunction photodetectors based on transition-metal dichalcogenides (TMDCs) such as  $\text{MoS}_2/\text{WSe}_2$   
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56 [21, 34] and  $\text{MoTe}_2/\text{MoS}_2$  [23]. The corresponding detectivity is estimated to be  $D^* = 3.7 \times 10^{12}$  Jones,  
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**Figure 3.** Band alignment at the interface of the  $p$ -GaSe/ $n$ -InSe heterojunction. (a) Band alignment for isolated  $n$ -InSe and  $p$ -GaSe layers. Electron affinities of InSe and GaSe are  $\chi = -4.6$  and  $-3.7$  eV, respectively. The conduction minimum (CB) of GaSe lies above that of InSe by  $\Delta E_c = 0.9$  eV whereas the valence band (VB) edge of InSe lies below ( $\Delta E_v = -0.1$  eV) that of the larger band gap GaSe, resulting in a type II band alignment. (b,c,d,e) Schematic band alignment at the interface of the heterojunction at different applied voltages  $V_{ds}$  (reverse bias (c), zero bias (b) and forward bias (d,e)).  $V_{FB}$  is the voltage corresponding to the flat band condition.

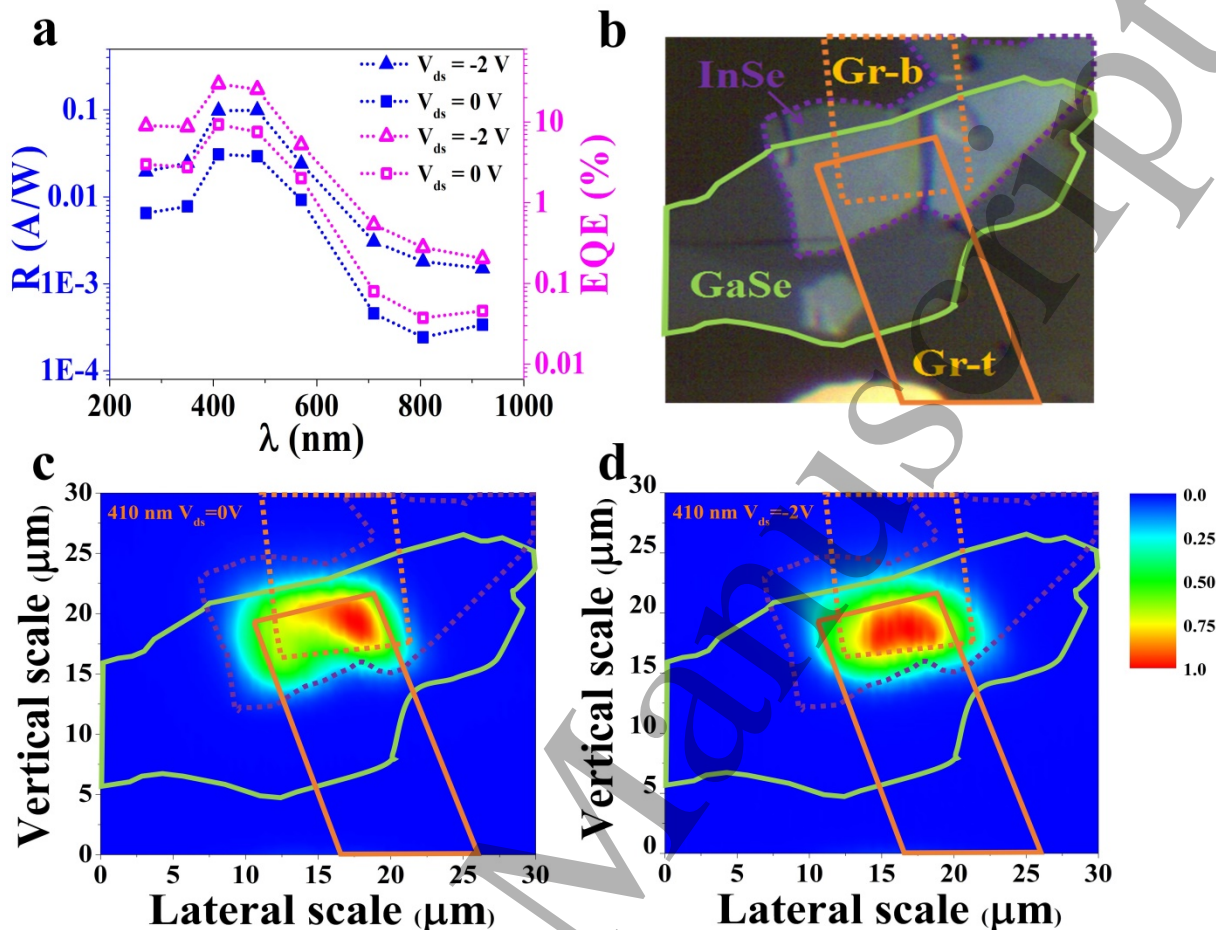
which is two orders of magnitude higher than that of InGaAs/InGaP-based [35] and MoS<sub>2</sub>-based [36] photodetectors, and is similar to that of Si  $p$ - $n$  junction photodetectors [37]. These high  $R$  and  $D^*$  values indicate that the  $p$ -GaSe/ $n$ -InSe vertical heterojunction is extremely sensitive to small optical input signals. Furthermore, these devices can operate with no externally applied voltage, thus they have potential for applications that require miniaturized devices with minimal energy consumption.

The spectral response of the  $p$ -GaSe/ $n$ -InSe heterojunction in figure 4(a) demonstrates a strong photoresponsivity over the range  $\lambda = 270$ - $920$  nm, from UV to NIR, under both reverse and zero biases.

The photoresponsivity in both reverse bias,  $V_{ds} = -2$  V, and zero bias display a similar wavelength

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4 dependence. The peak in the photoresponsivity between 400 and 500 nm corresponds to excitations  
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6 between the  $p_{xy}$ -like orbitals at the top of the valence band of GaSe and the minimum of the conduction  
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8 band of InSe. The UV response at  $\lambda = 270$  and 350 nm is due to interband optical absorption in the GaSe  
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10 layer, as for the case of GaSe-based photodetectors [13, 38]. The photoresponse in the NIR wavelength  
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12 range arises from interband transitions in the InSe layer, which has a smaller band gap of 1.26 eV at room  
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14 temperature [20]. To elucidate the role of graphene in our measurements, we have compared the  
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16 photocurrent spectra of GaSe- and InSe-based photodetectors with metal and graphene electrodes, see  
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18 new figure S3 in the supplementary information. A larger photoresponse was observed in photodetectors  
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20 based on graphene electrodes. Based on the photocurrent and incident laser power, we can determine the  
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22 external quantum efficiency,  $EQE$ , of the photon to electron conversion (figure 4(a)). The  $EQE (= hcR/e\lambda)$   
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24 is defined as the ratio of the number of carriers collected by the electrodes to the number of incident  
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26 photons, and is wavelength dependent. At zero bias, the device has a maximum  $EQE$  of 9.3% at  $\lambda = 410$   
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28 nm, higher than for monolayer MoS<sub>2</sub>/Si  $p$ - $n$  diodes [39].  
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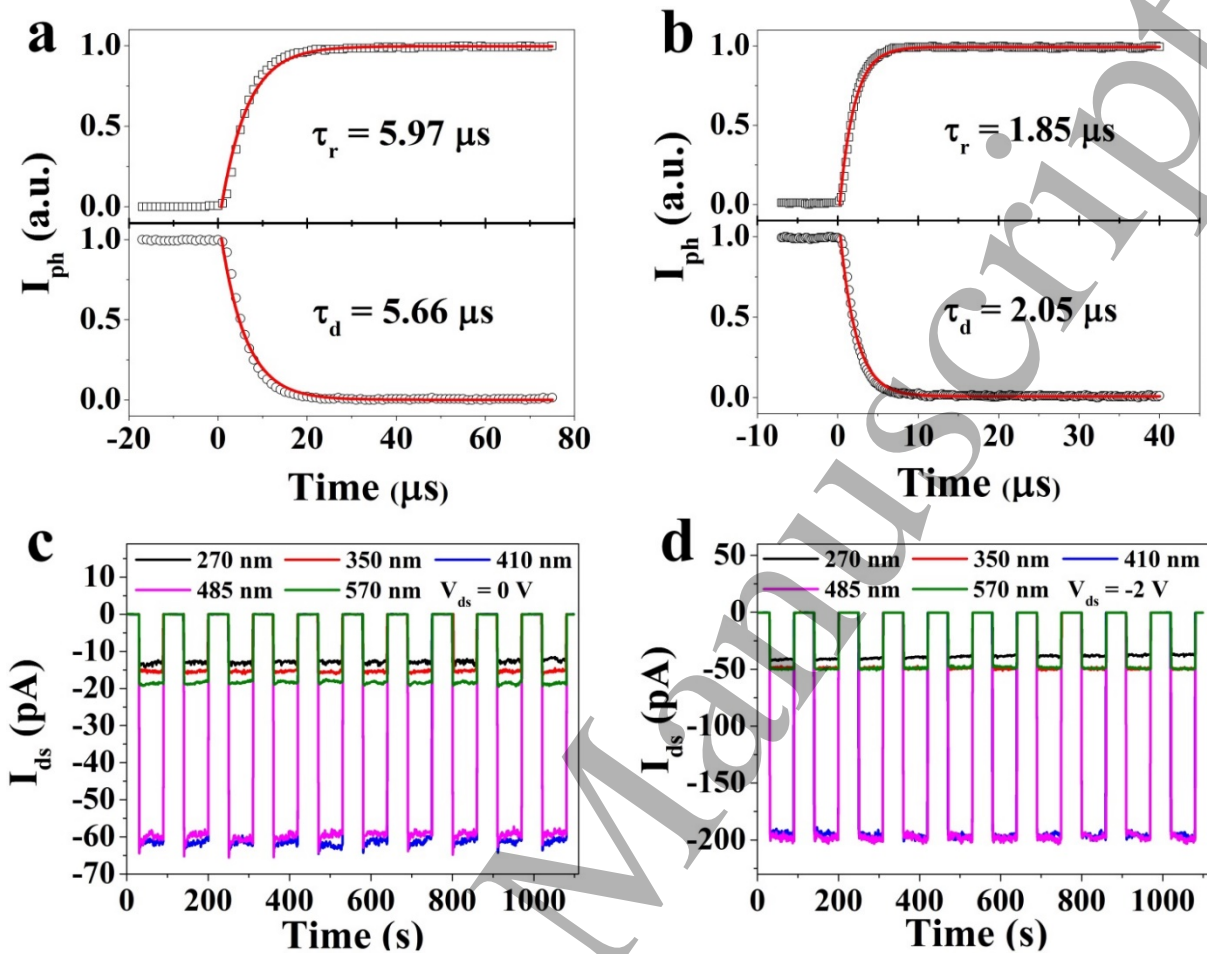
39 To further explore the origin of the photoresponse, photocurrent maps were acquired at both zero and  
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41 reverse biases. Figure 4(b) shows an optical microscope image of the Gr/GaSe/InSe/Gr heterostructure  
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43 depicting the relative positions of the GaSe and InSe layers and of the graphene electrodes. The  
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45 corresponding normalized photocurrent maps at zero and reverse biases with  $\lambda = 410$  nm laser excitation  
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47 (20  $\mu$ W) are shown in figures 4(c) and (d), respectively. To distinguish the different parts of the  
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49 heterostructure, the GaSe sheet region is outlined in green, the InSe sheet region in purple, and the top  
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51 and bottom graphene electrodes in solid and dotted gold lines, respectively. The photocurrent map shows  
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53 that the photosensitive region corresponds to the area where the four component layers  
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**Figure 4.** Spectral responsivity and normalized photocurrent maps of the  $p$ -GaSe/ $n$ -InSe heterojunction. (a) Room temperature photoresponsivity,  $R$ , and external quantum efficiency,  $EQE$ , as a function of illumination wavelength at different  $V_{ds}$  ( $V_{ds} = 0$  V and  $V_{ds} = -2$  V) and illumination intensity  $P = 1$  mW cm $^{-2}$ . The spot size of the laser beam is about 0.2 mm $^2$ . (b) Optical microscope image of the heterojunction; t and b refer to top and bottom graphene electrodes. (c,d) Normalized photocurrent maps of the Gr/GaSe/InSe/Gr device obtained by scanning a focused laser beam at  $V_{ds} = 0$  V (c) and  $-2$  V (d) with wavelength  $\lambda = 410$  nm ( $P = 20$   $\mu\text{W}$ ). The green solid line outlines the GaSe sheet, the purple dotted line outlines the InSe sheet and the golden solid and dotted lines outline the top and bottom graphene electrodes, respectively. Photocurrent is observed where the 4 layers overlap. The laser beam is focused by a microscope objective onto a spot size of diameter of about 1.5  $\mu\text{m}$ .

(Gr/GaSe/InSe/Gr) are superimposed and demonstrates the formation of a  $p$ - $n$  junction across the area of the GaSe/InSe interface and the efficient extraction of carriers into the graphene electrodes. The weak photocurrent from the non-overlapping regions shows that the photogenerated carriers in the regions outside the  $p$ - $n$  junction are separated and extracted less efficiently, even under a reverse bias  $V_{ds} = -2$  V (figure 4(d)).

The response time is another important indicator of the performance of a photodetector. To assess their behavior in the time domain, the devices were illuminated with pulsed light generated by a light-emitting diode (LED) ( $\lambda = 470$  nm) powered by a square-wave signal generator. As shown in figure 5(a), the dynamic response of the photocurrent at  $V_{ds} = 0$  V is described well by the equations  $I(t) = I_0[1 - \exp(-t/\tau_r)]$  and  $I(t) = I_0\exp(-t/\tau_d)$ , where  $\tau_r = 5.97$   $\mu$ s and  $\tau_d = 5.66$   $\mu$ s are the rise- and decay-time constants. The even faster photoresponse at  $V_{ds} = -2$  V with  $\tau_r = 1.85$   $\mu$ s and  $\tau_d = 2.05$   $\mu$ s (figure 5(b)) is due to the enhanced electric field in reverse bias. Figures 5(c) and (d) show that the photocurrent can be switched on and off repeatedly and reproducibly with a square-wave modulation of the light intensity for different laser wavelengths ( $\lambda = 270, 350, 410, 485,$  and  $570$  nm) at a power  $P = 1$  mW cm<sup>-2</sup>. Similar switching behavior is observed for NIR photo-excitation ( $\lambda = 920$  nm) under different illumination intensities for both zero and reverse biases (see Supplementary Information, figure S4). ON/OFF ratios up to  $10^3$  are observed, demonstrating that the heterojunction can be used as a sensitive and fast photo-detector. The measured response times are significantly faster than those recently reported for van der Waals heterojunction photodetectors [10, 19, 23, 24, 40] and Si-based heterojunction photodetectors [41, 42]. We also note that the measured response time is limited by the  $RC$ -time of our instrument and hence that a faster response could be obtained by improving the measuring circuit [43, 44].



**Figure 5.** Response time and photo-switching of the *p*-GaSe/*n*-InSe heterojunction diode. (a,b) Temporal dependence of the photocurrent and times  $\tau_r$  and  $\tau_d$  at  $V_{ds} = 0$  V (a) and  $V_{ds} = -2$  V (b) at room temperature. The red solid lines are fits to the data. (c,d) Source-drain current  $I_{ds}$  as a function of time with photoswitching at  $V_{ds} = 0$  V (c) and  $V_{ds} = -2$  V (d) under illumination with different wavelengths ( $\lambda = 270, 350, 410, 485,$  and  $570$  nm) and light intensity  $P = 1$  mW cm<sup>-2</sup>. The spot size of the laser beam is about 0.2 mm<sup>2</sup>.

## Conclusion

In conclusion, we have reported on a novel van der Waals multi-layer heterostructure that combines several two-dimensional van der Waals crystals, *i.e.* graphene and the metal monochalcogenide InSe and GaSe layered semiconductors. The latter present a number of features that distinguish them from the widely explored TMDCs. In particular, they have a type II band alignment and have distinctive spectral response. We have exploited these features and the low electrical resistance and optical transparency of monolayer graphene electrodes to fabricate a junction diode that can adsorb light over a broad spectral range, from the ultraviolet, visible and infrared, and in which the photo-generated carriers can be efficiently and quickly extracted from the InSe/GaSe heterostructure even when no external voltage is applied. The low or zero energy consumption, simple heterostructure design and fast response time down to  $\sim 1 \mu\text{s}$  are notable important features of these nanometer-scale devices. Our results will stimulate further research into the science and technology of these heterostructures, which have the potential for a wide range of applications.

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