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# Manipulating adsorbed hydrogen on lanthanum-modified $CuO_x$ : Industrial-current-density $CO_2$ electroreduction to $C_{2+}$ products or $CH_4$

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# ABSTRACT

The selective electrochemical CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) yields valuable C<sub>2+</sub> and C<sub>1</sub> products, yet the influence of adsorbed hydrogen (\*H) on product distribution remains inadequately understood. This study explores this effect by developing bimetallic copper-based electrocatalysts with varied lanthanum (La) doping ratios. The oxide-derived (OD)-La<sub>0.10</sub>-CuO<sub>x</sub> catalyst exhibits a Faradaic efficiency (FE) over 80% for C<sub>2+</sub> products at 300 mA cm<sup>-2</sup>, whereas OD-La<sub>0.40</sub>-CuO<sub>x</sub> achieves a 61.4% FE<sub>CH4</sub> at 400 mA cm<sup>-2</sup>. Kinetic isotope experiments reveal distinct dependencies of the rate-determining steps on \*H transfer for CO<sub>2</sub>RR in OD-La<sub>0.10</sub>-CuO<sub>x</sub> and OD-La<sub>0.40</sub>-CuO<sub>x</sub> lowers the energy barrier for \*CHO  $\rightarrow$  \*OCCHO conversion, thus increasing the FE<sub>C2+</sub>. Conversely, OD-La<sub>0.40</sub>-CuO<sub>x</sub>, with its strong H<sub>2</sub>O dissociation capability, favors \*CHO  $\rightarrow$  \*CH<sub>2</sub>O, thereby promoting CO<sub>2</sub>RR-to-CH<sub>4</sub>. These findings advance the understanding of the role of \*H in CO<sub>2</sub> electroreduction at industrial current densities and present avenues for tailored CO<sub>2</sub>RR products via doping engineering.

#### 1. Introduction

The direct electrochemical carbon dioxide reduction reaction  $(CO_2RR)$ , driven by renewable energy, represents a promising approach for achieving sustainable carbon-neutral energy conversion [1,2]. This environmentally friendly strategy enables carbon recovery while simultaneously producing valuable chemicals and fuels [3,4]. Among the various catalysts explored for CO<sub>2</sub>RR, copper (Cu)-based catalysts facilitate CO<sub>2</sub> electroreduction toward multicarbon products at industrially relevant current densities (>200 mA cm<sup>-2</sup>) [5]. This capacity is attributed to copper's negative adsorption energy for adsorbed CO (\*CO) and positive adsorption energy for adsorbed hydrogen (\*H), enabling complex coupling processes involving multiple proton and electron transfers [6]. CO<sub>2</sub>RR products such as methane (CH<sub>4</sub>), a 6-electron product, and ethylene (C<sub>2</sub>H<sub>4</sub>), a 12-electron product, are commonly used as high energy density fuels or high value-added chemicals [7–13]. Despite extensive research on high-performance Cu-based electrocatalysts, the structure-function relationships in CO<sub>2</sub>RR and the reaction pathways to various products at industrial current densities are not fully understood. A better understanding of correlations among various products would offer the prospect of increasing the conversion efficiency of desired CO<sub>2</sub>RR products.

In CO<sub>2</sub>RR, multiple proton and electron transfer steps occur on the electrocatalyst surface, either sequentially or concurrently, leading to uncontrollable product distributions [14,15]. Researchers have elucidated CO<sub>2</sub>RR selectivity using theoretical and experimental approaches. For the C1 product formation, Liu et al. demonstrated that the rate-determining step (RDS) in the CO<sub>2</sub>RR-to-CO pathway is the coupling of  $*CO_2^-$  with H<sup>+</sup> to form \*COOH [16]. Xiong et al. revealed that the hydrogenation of \*CO to \*CHO is the RDS in the CO2RR-to-CH4 pathway, observing the \*CHO intermediate using in situ Fourier transform infrared spectroscopy (FTIR) [17]. Toward C<sub>2+</sub> hydrocarbon formation, C-C coupling is widely recognized as a crucial step. Kim and collaborators observed the formation of \*OCCO species through C-C coupling during the CO<sub>2</sub>RR-to-C<sub>2</sub> process at a Cu electrode, using in situ attenuated total reflectance-surface-enhanced infrared absorption spectroscopy (ATR-SEIRAS) [18,19]. An increased local \*CO concentration facilitates further protonation of \*CO-\*CO coupling, progressing

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to \*CHO-\*CO coupling [20]. These studies highlight that proton/\*H activation and transfer are integral to various  $CO_2RR$  pathways, particularly for synthesizing desirable hydrocarbons such as  $CH_4$ ,  $C_2H_4$ , and other  $C_{2+}$  products.

Typically, protons involved in CO<sub>2</sub>RR are derived from sources such as  $H_2O$ ,  $H_3O^+$  and  $HCO_3^-$  [21]. An isotope tracing study has demonstrated the direct role of H<sub>2</sub>O in the electrochemical reduction of CO to ethanol [22]. Moreover, it is reported that accelerated H<sub>2</sub>O dissociation kinetics can lower the energy barrier for formate formation [23]. Therefore, customizing H<sub>2</sub>O dissociation to modulate local \*H donors emerges as a promising strategy to facilitate CO2RR, enabling tailored distributions of C1 and C2+ products. The introduction of single-atom [24,25], oxygen vacancies [23], and rare-earth element doping, such as lanthanide [26], on catalyst surfaces has been reported to provide sites for H<sub>2</sub>O dissociation. These sites expedite H<sub>2</sub>O dissociation into \*H, which subsequently participates in forming various hydrocarbons at CO<sub>2</sub> conversion sites [21,22,27]. Consequently, hydrocarbon production strongly depends on H<sub>2</sub>O dissociation. Among these strategies for modulating \*H, heteroatom doping can alter the charge density around active sites and affect reaction intermediates, thereby improving the activity and selectivity of Cu-based catalysts for CO<sub>2</sub>RR [28]. Compared to d-block metal elements, the unique physical and chemical properties of lanthanides stem from the special positioning of their 4 f electrons and partially filled 5d orbitals. Specifically, lanthanum exhibits a high reduction potential ( $La^{3+}/La^{0}$ , -2.90 V vs RHE), remains stable during the CO2RR process, and thus serves to modify the electronic structure of the Cu substrate [29]. For instance, La has been incorporated into constructs like La(OH)3/Cu, Cu-based perovskite oxides, and Cu-based allovs for  $CO_2$  electroreduction to various products [7,29,30].

While previous research has focused primarily on \*H's effect on single product formation through  $CO_2RR$ , there is limited research on the branching pathways of  $CO_2RR$  to  $C_1$  and  $C_{2+}$  products and their relationship with \*H, particularly regarding \*H transfer mechanisms in hydrogenation processes and competition with other intermediate formation processes such as dimerization. Although alkaline electrolytes commonly employed in industrial-current-density  $CO_2$  electroreduction hinder H<sub>2</sub>O dissociation and \*H transfer, thus effectively inhibiting the hydrogen evolution reaction (HER), they also result in sluggish  $CO_2RR$ kinetics due to delayed \*H activation and transfer [31,32]. Therefore, managing adsorbed hydrogen production and its transfer kinetics, and simultaneously suppressing HER is crucial for the rational design of high-performance  $CO_2RR$  electrocatalysts.

In this study, we synthesized a series of oxide-derived lanthanummodified copper oxide (denoted as OD-La-CuOx) catalysts with varied La doping to investigate the effect of \*H on CO2RR product distribution at industrial current densities. The results show that hydrocarbon selectivity shifts from C<sub>2+</sub> products (in OD-La<sub>0.10</sub>-CuO<sub>x</sub>) to CH<sub>4</sub> (in OD-La<sub>0.40</sub>-CuO<sub>x</sub>) as La doping increases. Kinetic isotope effect (KIE) experiments and in situ electrochemical impedance spectroscopy (EIS) were employed to elucidate the role of \*H transfer in forming various CO2RR products. In situ ATR-SEIRAS and density functional theory (DFT) calculations reveal that La doping promotes H<sub>2</sub>O adsorption and its dissociation to release \*H. Additionally, the divergence between C2+ and CH<sub>4</sub> formation pathways lies in the subsequent conversion of \*CHO. Specifically, during the CO<sub>2</sub>RR-to-C<sub>2+</sub> process, OD-La<sub>0.10</sub>-CuO<sub>x</sub> facilitates \*CO protonation to \*CHO and its further dimerization to \*OCCHO, whereas OD-La<sub>0.40</sub>-CuO<sub>x</sub> promotes \*CHO hydrogenation to \*CH<sub>2</sub>O, leading to CH<sub>4</sub> production.

#### 2. Experimental section

# 2.1. Synthesis of La-CuO<sub>x</sub> catalysts

The La<sub>0.10</sub>-CuO<sub>x</sub> catalyst was synthesized using a facile alcoholthermal approach. Initially, Cu(CH<sub>3</sub>COO)<sub>2</sub>·H<sub>2</sub>O (3 mmol, 0.6 g) and La (NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O (0.3 mmol, 0.13 g) were dissolved in 50 mL ethanol through sonication for 30 minutes with the lanthanum to copper molar ratio of 0.1–1. The resultant blue mixture was transferred into a Teflonlined stainless-steel autoclave with a volume of 100 mL, and heated at 150 °C for 20 hours in an oven. After allowing the system to cool to room temperature naturally, the precipitates were washed three times with ethanol and deionized water before being dried at 70 °C overnight. Additionally, CuO<sub>x</sub> was synthesized under identical conditions without the inclusion of La(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O. Various La-modified CuO<sub>x</sub> catalysts (La<sub>0.40</sub>- CuO<sub>x</sub>, La<sub>0.20</sub>-CuO<sub>x</sub>, La<sub>0.13</sub>-CuO<sub>x</sub>, and La<sub>0.06</sub>-CuO<sub>x</sub>) were prepared using a similar procedure, with adjustments in La-Cu ratios such that 1.2 mmol, 0.6 mmol, 0.4 mmol, and 0.2 mmol of La(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O were added, respectively. The as-prepared catalysts was transformed into OD-La-CuO<sub>x</sub> through electroreduction.

# 2.2. Preparation of electrodes

A commercially available carbon paper (YLS-30T), utilized as a gas diffusion electrode (GDE), was precisely cut to dimensions of 2.0 cm  $\times$  15.0 cm and thoroughly rinsed with deionized water. A catalyst weighing 30 mg, such as CuO<sub>x</sub>, La<sub>0.06</sub>-CuO<sub>x</sub>, La<sub>0.10</sub>-CuO<sub>x</sub>, La<sub>0.13</sub>-CuO<sub>x</sub>, La<sub>0.20</sub>-CuO<sub>x</sub>, or La<sub>0.40</sub>-CuO<sub>x</sub>, was dispersed in a mixture containing 20  $\mu$ L of Nafion solution (5 wt%) and 3 mL of isopropanol, followed by sonication for over an hour. This prepared catalyst slurry was sprayed onto the cleaned GDE using a handheld airbrush, achieving a catalyst loading of approximately 1 mg cm<sup>-2</sup>. After drying the cathode electrodes in a vacuum oven at 60 °C for 10 minutes, they were sectioned into seven squares, each measuring approximately 2.0 cm  $\times$  2.0 cm.

#### 2.3. Electrochemical performance tests

The electrochemical performance tests were conducted in a gas-fed flow cell configurated with either an anion exchange membrane (AEM, Fumasep FAA-3-PK-130) or a cation exchange membrane (CEM, Nafion 117), depending on the electrolyte used (1 M KOH and 1 M KCl) to satisfy the industrial requirement. The electrolyzer comprised three chambers: the airflow chamber (1.0 cm  $\times$  1.0 cm), the cathode chamber with the working electrode (2.0 cm  $\times$  2.0 cm) and an Ag/AgCl reference electrode (saturated with 3 M KCl), and the anode chamber containing nickel foam (2.0 cm  $\times$  2.0 cm). The active area of the catholyte chamber was  $1.0 \text{ cm}^2$ . A high-purity CO<sub>2</sub> gas stream (99.99%) was passed through the gas chamber at a rate of 50 sccm, controlled by a mass flow meter. The cathode chamber was supplied with 10 mL min<sup>-1</sup> of 1 M KOH (for alkaline condition, pH = 13.6) and 1 M KCl (for neutral condition, pH =6.4) aqueous electrolytes via a peristaltic pump, while the anode chamber was fed with 10 mL min<sup>-1</sup> 1 M KOH aqueous electrolyte using another peristaltic pump. The gas flow rate between the gas-fed flow cell and the GC (Agilent 8890B) inlet was verified by a soap film flow meter (Sensidyne Gilibrator-2).

All electrolysis experiments were conducted using an electrochemical workstation (CHI 660E) equipped with a current booster (CHI 680D). Potentials were converted to values relative to the reversible hydrogen electrode (RHE) with manual  $iR_u$  compensation, following the equation below:

# E (vs. RHE) = E (vs. Ag/AgCl) + 0.0591 × pH + 0.210 - $iR_u$ × 0.85

Cyclic voltammetry (CV) scans were performed prior to each CO<sub>2</sub>RR experiment to pre-activate the electrocatalyst and remove organic residues present on the electrode surface, using 20 cycles over a potential range of -0.6 to -2.0 V relative to Ag/AgCl. Gas products (H<sub>2</sub>, CO, CH<sub>4</sub> and C<sub>2</sub>H<sub>4</sub>) were monitored in real-time by connecting a GC inlet directly to the cathode chamber. The GC system is equipped with two flame ionization detectors (FID, the back FID for CO and CH<sub>4</sub>; the front FID for C<sub>2</sub>-C<sub>3</sub> hydrocarbons) and a thermal conductivity detector (TCD for H<sub>2</sub>) for the identification and quantitative analysis of these products. Faradaic efficiencies of the gas products were determined using Eq. S1 from

the raw GC data. Liquid products were quantified using proton nuclear magnetic resonance (<sup>1</sup>H NMR, Bruker DPX 400 MHz) with a water suppression technique. An internal standard was prepared by mixing 150  $\mu$ L of dimethyl sulfoxide (DMSO) with 300 mL of ultrapure water. The NMR test solution consisted of 500  $\mu$ L electrolyte, 100  $\mu$ L D<sub>2</sub>O, and 100  $\mu$ L of aqueous DMSO (containing 0.05  $\mu$ L DMSO) (Eq. S2 and Fig. S1). Cathodic energy efficiency was computed based on the CO<sub>2</sub>RR at the cathode coupled with water oxidation reaction at the anode following Eq. S3. The Tafel slope for CO<sub>2</sub>RR was calculated according to Eq. S4.

Additional details of the chemicals used, catalyst characterizations, product analysis, *in situ* ATR-SEIRAS measurements, and DFT computational methods are provided in the Supporting Information.

# 3. Results and discussion

## 3.1. Synthesis and characterizations

The lanthanum-modified  $\text{CuO}_x$  nanoparticles, synthesized with various La-doping molar ratios, were used as pre-electrocatalysts. As depicted in Fig. 1a, these catalysts were produced using a one-pot alcohol-thermal approach. The molar ratio of Cu to La was controlled by adjusting the concentration of the La precursor solution. Specifically, the La content for La<sub>0.10</sub>- CuO<sub>x</sub> and La<sub>0.40</sub>- CuO<sub>x</sub> was measured at 9.9 wt % and 25.6 wt%, respectively, by inductively coupled plasma mass spectrometry (ICP-MS) (Table S1). This facile synthesis method involves the decomposition of Cu(CH<sub>3</sub>COO)<sub>2</sub>·H<sub>2</sub>O and La(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O into mixed-phase metal oxide catalysts in the presence of ethanol. Under alcohol-thermal conditions within a Teflon-lined stainless-steel

autoclave, an esterification reaction between acetate and ethanol readily occurs. Previous research has confirmed the formation of ethyl acetate in reaction solvents through gas chromatography (GC) with a flame ionization detector (FID) [33].

Transmission electron microscopy (TEM) was employed to examine the structural morphology of the as-prepared La-modified CuOx electrocatalysts. The  $CuO_x$  nanoparticles, synthesized without La (NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O, display regular particle sizes of under 10 nm (Fig. S2). As shown in Fig. 1b and S3, low-level La doping (i.e., La<sub>0.10</sub>- CuO<sub>x</sub>) has a minimal impact on morphology, maintaining an average particle size of roughly 6 nm. However, with increased La doping (i.e., La<sub>0.40</sub>-CuO<sub>x</sub>), the original particle morphology is not preserved. Fig. 1c and S4 show that ultra-small nanoparticles tend to agglomerate and increase in size. High-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) images and corresponding energy-dispersive Xray spectroscopy (EDS) element maps reveal a uniform distribution of O, Cu, and La elements (Fig. 1b and c, Figs. S5 and S6) and indicate the integration of La into the fabricated material. Additionally, highresolution TEM (HRTEM) was applied to reveal the lattice fringe characteristics of the synthesised La- CuO<sub>x</sub> catalysts. The HRTEM images of both La<sub>0 10</sub>- CuO<sub>x</sub> and La<sub>0 40</sub>- CuO<sub>x</sub> exhibit abundant grain boundaries, with lattice edges in multiple orientations (Figs. S7 and S8). The grain boundaries of La<sub>0.10</sub>-CuO<sub>x</sub> are highlighted by white dotted lines in Fig. S7, resulting from the incorporation of heteroatoms into the material via doping. This introduction disrupts the ordered lattice structure and crystal growth, leading to defect formation and high-density grain boundaries within the material [26].

To verify the phase composition of the synthesized La-CuO<sub>x</sub> catalysts, powder X-ray diffraction (PXRD) analyses were performed on both the



Fig. 1. a) Schematic illustration of the synthetic procedures of CuO<sub>x</sub> and La-CuO<sub>x</sub> catalysts. TEM image, HAADF-STEM image, and corresponding EDS element maps of b) La<sub>0.10</sub>- CuO<sub>x</sub> and c) La<sub>0.40</sub>- CuO<sub>x</sub>.

pure CuO<sub>x</sub> and La-CuO<sub>x</sub> catalysts (Fig. 2a). For CuO<sub>x</sub> without La doping, the diffraction peaks can be precisely indexed to the monoclinic pure phase of CuO (JCPDS# 45-0937) with the most intense peaks at 35.6° for CuO (002) and 38.7° for CuO (111). When La is incorporated in small amounts, from La<sub>0.06</sub>-CuO<sub>x</sub> to La<sub>0.10</sub>-CuO<sub>x</sub>, the XRD patterns show a composition of CuO and Cu<sub>2</sub>O (JCPDS# 05-0667). As La-doping increases, the diffraction peaks of CuO (002) and (111) weaken, while the peak for Cu<sub>2</sub>O (111) becomes more pronounced. This behavior is attributed to the ability of trivalent La to modify the CuO electronic structure, initiating possible redox reactions [26], such as the replacement of Cu<sup>2+</sup>-O<sup>2-</sup>-Cu<sup>2+</sup> pairs with Cu<sup>+</sup>-O<sup>2-</sup>-La<sup>3+</sup> pairs to maintain electroneutrality [34]. With significant La doping, particularly from La<sub>0.13</sub>-CuO<sub>x</sub> to La<sub>0.40</sub>-CuO<sub>x</sub>, additional crystalline phase is observed, attributed to La<sub>2</sub>CuO<sub>4</sub> (as per JCPDS# 30–0487), with increasing peak intensity. Concurrently, the diffraction peaks of CuO and Cu<sub>2</sub>O diminish, with some eventually disappearing. Notably, the Cu<sub>2</sub>O (111) and CuO ( $\overline{113}$ ) facet peaks of La-CuO<sub>x</sub> with higher La doping levels are slightly shifted to lower diffraction regions compared to La<sub>0.06</sub>-CuO<sub>x</sub> and undoped CuO<sub>x</sub>, respectively (Fig. S9). This shift is attributed to the doped La atoms, which have a larger atomic radius than Cu atoms [35, 36]. The Raman spectra of CuO<sub>x</sub> show three peaks at 285, 336, and 619 cm<sup>-1</sup> (Fig. S10), which correspond to the A<sub>g</sub>, B<sub>g1</sub> and B<sub>g2</sub> vibration modes of CuO, respectively. The Raman peaks for La<sub>0.10</sub>-CuO<sub>x</sub> are shifted to higher wavenumbers, indicating changes in the electronic structure of CuO due to La doping. Moreover, a new Raman peak at 217 cm<sup>-1</sup> is assigned to the  $2\Gamma_{12}$  phonon modes of the Cu<sub>2</sub>O [35,36].

Following CO<sub>2</sub>RR at a current density of 300 mA cm<sup>-2</sup> in 1 M KCl for 30 minutes, the OD-CuO<sub>x</sub> only displays the characteristic peaks of the metallic Cu phase (Fig. S11a). In contrast, the OD-La<sub>0.10</sub>-CuO<sub>x</sub> retains peaks corresponding to the Cu<sub>2</sub>O phase, dominated by the (111) facets, even after 60 minutes (Figs. 2b, S11b and S12). This is likely due to the construction of unconventional 4f-2d-3d hybrid orbitals through La



**Fig. 2.** a) XRD patterns of  $CuO_x$  and various La- $CuO_x$  catalysts prior to  $CO_2RR$ . The XRD patterns of b) OD-La<sub>0.10</sub>- $CuO_x$  and c) OD-La<sub>0.40</sub>- $CuO_x$  after  $CO_2RR$  at 300 mA cm<sup>-2</sup> for 30 minutes in 1 M KCl and 1 M KOH, respectively. d) Normalized intensity and e) First-order derivative of Cu K-edge X-ray absorption near edge structure (XANES spectra for La<sub>0.10</sub>-CuO<sub>x</sub>, La<sub>0.40</sub>-CuO<sub>x</sub> and other reference standards. High-resolution XPS spectra of f) Cu 2p, g) La 3d, and h) O 1 s of CuO<sub>x</sub> and the as-prepared La-CuO<sub>x</sub> catalysts.

doping, which has been reported to effectively prevent lattice oxygen leaching, thereby stabilizing Cu<sup>+</sup> species and maintaining the Cu<sub>2</sub>O phase [37-40]. The diffraction patterns of OD-La<sub>0.40</sub>-CuO<sub>x</sub> were also examined at various electrolysis duration in 1 M KOH at a current density of 400 mA cm $^{-2}$ . The crystalline phases in OD-La<sub>0.40</sub>-CuO<sub>x</sub>, comprising La<sub>2</sub>CuO<sub>4</sub> and Cu<sub>2</sub>O, remain stable throughout electrolysis, although the proportion of the Cu<sub>2</sub>O phase decreases after 30 minutes (Figs. 2c and S11c). Additionally, ex situ Cu LMM Auger spectra of the reconstructed OD-CuOx, OD-La<sub>0.10</sub>-CuOx, and OD-La<sub>0.40</sub>-CuOx were gathered under identical conditions as ex situ XRD tests to elucidate the Cu oxidation states. During CO<sub>2</sub>RR, OD-CuO<sub>x</sub> shows an increasing proportion of Cu<sup>0</sup> species (918.4 eV) over 10-min increments, which fully occupies by 30 and 60 minutes (Fig. S13a). Conversely, the Cu LMM spectra for OD-La<sub>0.10</sub>-CuO<sub>x</sub> indicate a stabilization of Cu<sup>+</sup> species (916.8 eV) during 60 minutes of CO<sub>2</sub> electrolysis (Fig. S13b). Furthermore, the spectra for OD-La<sub>0.40</sub>-CuO<sub>x</sub> reveal a mixture of Cu<sup>+</sup> (916.8 eV) and Cu<sup>2+</sup> (917.8) species both before and after various electrolysis intervals (Fig. S13c). Notably, the ex situ X-ray photoelectron spectroscopy (XPS) of La 3d spectra reveals that La<sup>3+</sup> species remain unchanged in both OD-La<sub>0.10</sub>-CuO<sub>x</sub> and OD-La<sub>0.40</sub>-CuO<sub>x</sub> during 60 minutes of CO<sub>2</sub> electrolysis. As illustrated in Fig. S14, peaks at 838.8 eV and 855.6 eV correspond to  $La^{3+} 3d_{5/2}$  and  $3d_{3/2}$ , respectively. The spin-orbit splitting energy of 16.8 eV between the  $3d_{5/2}$  and  $3d_{3/2}$  states of La<sup>3+</sup> is consistent with reported values for La<sub>2</sub>O<sub>3</sub> [41].

To further identify the Cu oxidation state and microstructure of localized Cu species, X-ray absorption spectroscopy (XAS) was conducted at the Cu K-edge for La<sub>0.10</sub>-CuO<sub>x</sub>, La<sub>0.40</sub>-CuO<sub>x</sub> and reference standards (Cu, Cu<sub>2</sub>O and CuO). As shown in Fig. 2d, the absorption edges of La<sub>0.10</sub>-CuO<sub>x</sub> and La<sub>0.40</sub>-CuO<sub>x</sub> in the normalized Cu K-edge X-ray absorption near edge structure (XANES) are positioned between those of the Cu<sub>2</sub>O and CuO standards, indicating Cu valence states between +1 and +2 for both samples. This finding aligns with XRD results, showing that  $La_{0.10}$ -CuO<sub>x</sub> and  $La_{0.40}$ -CuO<sub>x</sub> possess mixed phases of Cu<sub>2</sub>O, CuO and La2CuO4. Furthermore, the white-line intensity of La0.40-CuOx surpasses that of the CuO standard and  $La_{0.10}$ -CuO<sub>x</sub>, implying a higher oxidation state of Cu in  $La_{0.40}$ -Cu. The normalized first-order derivative of XANES for La<sub>0.10</sub>-CuO<sub>x</sub> and La<sub>0.40</sub>-CuO<sub>x</sub> display peaks overlapping with Cu<sub>2</sub>O and CuO at approximately 8980 eV and 8984 eV, verifying the existence of Cu(I) and Cu(II) species in La<sub>0.10</sub>-CuO<sub>x</sub> and La<sub>0.40</sub>-CuO<sub>x</sub> (Fig. 2e). In conjunction with XRD analysis, the Cu(II) species in  $La_{0.40}$ -CuO<sub>x</sub> mainly derive from the La<sub>2</sub>CuO<sub>4</sub> phase.

High-resolution XPS was employed to investigate electronic interactions between Cu and La and to determine the surface chemical states of elements in the as-prepared catalysts. As expected, the full XPS spectra confirmed the presence of Cu, La and O, aligning with the EDS analysis of the La-CuO<sub>x</sub> catalysts (Fig. S15). With increased La doping, there is a marked reduction in peak intensity of Cu 2p, whereas that for La 3d rises. Fig. 2f depicts the core-level XPS spectra of Cu 2p for CuO<sub>x</sub>, La<sub>0.10</sub>-CuO<sub>x</sub>, and other La-CuO<sub>x</sub> catalysts with different La doping levels. Notably, the Cu  $2p_{3/2}$  and Cu  $2p_{1/2}$  peaks in the La-CuO<sub>x</sub> catalysts shift to higher binding energies with increased La doping compared to CuO<sub>x</sub> peaks. This suggests charge transfer from Cu to La, attributed to the unique electronic structure of La's unfilled 4 f orbitals [40]. Furthermore, strong Cu<sup>2+</sup> satellite peaks, present in all samples, range between 940 eV and 945 eV, indicating either CuO or La2CuO4 phases, as supported by XRD analysis. Additionally, the peaks located between 917.8 eV and 916.8 eV in the Cu LMM spectra confirm the presence of Cu<sup>2+</sup> and Cu<sup>+</sup> species (Fig. S16). This finding aligns with the XRD results (Fig. 2a) discussed earlier. Similarly, the core-level XPS spectra of La 3d (Fig. 2g) show significant shifts of La  $3d_{3/2}$  and La  $3d_{5/2}$  peaks to lower energy with increased La content, suggesting electron transfer from Cu to La, which corroborates the findings from the Cu 2p XPS analysis and highlights favorable charge transfer within the La-CuO<sub>x</sub> catalyst system.

To further explore the distribution of oxygen species in  $La-CuO_x$  catalysts, the O 1 s spectra of the as-prepared  $La-CuO_x$  catalysts were examined via XPS. As shown in Fig. 2h, the asymmetric O 1 s peaks

indicate the presence of different oxygen species, which deconvolute into three peaks at binding energies of 529.7, 531.5 and 532.7 eV, respectively. The peak at 529.7 eV corresponds to the lattice oxygen ( $O_L$ ) within the Cu-O-Cu and Cu-O-La bonds [42,43]. The peaks at 531.5 and 532.7 eV are attributed to surface-adsorbed hydroxyl or oxygen species ( $-OH/O_2$ ,  $O_{ads}$ ) and surface-adsorbed H<sub>2</sub>O on the La-CuO<sub>x</sub> matrix, respectively [44,45]. Comparing the  $O_{ads}$  to  $O_L$  ratio reveals La doping markedly increases the proportion of surface-adsorbed  $-OH/O_2$ , elevating the ratio from 0.6 to 3.5. XPS results demonstrate that the La-CuO<sub>x</sub> catalysts, fabricated through this one-pot thermal process, are rich in surface-adsorbed hydroxyl or oxygen species. Prior studies indicate that surface oxygen-containing species play a role in stabilizing reaction intermediates for enhanced  $CO_2RR$  performance [46].

#### 3.2. Electrochemical performance of CO<sub>2</sub>RR

The electrocatalytic CO<sub>2</sub>RR performance of oxide-derived CuO<sub>x</sub> and La-CuO<sub>x</sub> catalysts was evaluated in a custom liquid electrolyte flow cell under constant current densities in the range of 100–500 mA cm<sup>-2</sup> (Fig. S17). Unless otherwise specified, potentials were converted to the reversible hydrogen electrode (RHE) scale using an 85% *iR* correction to minimize the variations between the actual and desired potential at high current densities. The catholyte consisted of 1 M KOH and 1 M KCl solutions to evaluate the electrocatalysts' performance at industrial current densities in both alkaline and neutral electrolytes, respectively. Prior to each on-line GC test, air-tightness checks, cyclic voltammetry (CV) activation, and gas outlet flow rate tests were performed.

To assess the selectivity towards C<sub>2</sub>H<sub>4</sub>, Faradaic efficiencies (FEs) for the synthesized OD-CuOx and La<sub>0.10</sub>-CuOx were evaluated in alkaline and neutral electrolytes across current densities of 100 to 500 mA cm<sup>-2</sup>. As shown in Fig. 3a, the CuO<sub>x</sub> catalyst with its ultrasmall particle size exhibits a gradual increase in C2H4 selectivity with rising cathodic current density, achieving a maximum FE<sub>C2H4</sub> of about 40%. In contrast, the OD-La<sub>0.10</sub>-CuO<sub>x</sub> shows enhanced FE<sub>C2H4</sub> in both electrolyte types across the same current density range. It achieves up to 52%  $\mathrm{FE}_{\mathrm{C2H4}}$  in 1 M KCl at a current density of 300 mA  $cm^{-2}$ . The use of neutral electrolytes significantly mitigates challenges related to (bi)carbonate precipitation and gas diffusion electrode (GDE) collapse, thus enabling efficient and stable electrolysis at high current densities [47,48]. To investigate the impact of La doping on electron selectivity in CO2RR-to-C<sub>2</sub>H<sub>4</sub>, chronopotentiometry tests were conducted on CuO<sub>x</sub> and La-CuOx catalysts in both 1 M KOH and KCl solutions at a current density of 300 mA cm<sup>-2</sup>. Fig. 3b illustrates a notable increase in FE<sub>C2H4</sub> with greater La content, peaking at OD-La<sub>0.10</sub>-CuO<sub>x</sub>. CO<sub>2</sub>RR-to-C<sub>2</sub>H<sub>4</sub> performance diminishes sharply with further La content increase, as observed in OD-La<sub>0.13</sub>-CuO<sub>x</sub> and beyond. Notably, for OD-La<sub>0.40</sub>-CuO<sub>x</sub>, FE<sub>C2H4</sub> drops to 1.5% and 1.7% in 1 M KOH and 1 M KCl, respectively (Fig. S18 and S19). This electron selectivity towards CH<sub>4</sub> can be attributed to the emergence of La<sub>2</sub>CuO<sub>4</sub> phase, progressively replacing the initial mixed phase of OD-La<sub>0 10</sub>-CuO<sub>x</sub> (Fig. 2a), aligning with the finding reported in a previous work [43]. Detailed gas product distribution data for each sample at different current densities and in different electrolytes, along with the Faradaic efficiency of gas and liquid products for OD-La<sub>0.10</sub>--CuOx in 1 M KCl and OD-La0.40-CuOx in 1 M KOH at current densities ranging from 100 to 500 mA  $cm^{-2}$ , are provided in Tables S2-S6.

To gain insights into  $C_{2+}$  product formation selectivity, the focus is placed on OD-La<sub>0.10</sub>-CuO<sub>x</sub>, which demonstrates the highest CO<sub>2</sub>RR-to-C<sub>2</sub>H<sub>4</sub> selectivity in 1 M KCl at 300 mA cm<sup>-2</sup>. The FEs for various gas/ liquid products over OD-La<sub>0.10</sub>-CuO<sub>x</sub> are compared in Fig. S20 to evaluate CO<sub>2</sub>RR-to-C<sub>2+</sub> performance. Even at high current densities (300 mA cm<sup>-2</sup>), OD-La<sub>0.10</sub>-CuO<sub>x</sub> achieves an FE exceeding 80% for C<sub>2+</sub> products (Fig. 3c) via CO<sub>2</sub> electroreduction, comprising 51.8% FE<sub>C2H4</sub>, 21.5% FE<sub>C2H5OH</sub>, and 7.1% FE<sub>n-propanol</sub> (Fig. S20 and Table S4). This impressive performance is attributed to the suppression of the competing HER. The FE of HER is limited to approximately 10% at current densities between 200 and 400 mA cm<sup>-2</sup> (Fig. S20). Moreover,



**Fig. 3.** Performance on  $CO_2RR$  to  $C_{2+}$  products and  $CH_4$  in a flow cell. a) FE of  $C_2H_4$  on OD-La<sub>0.10</sub>-CuO<sub>x</sub> and OD-CuO<sub>x</sub> in 1 M KOH and 1 M KCl at current densities from 100 to 500 mA cm<sup>-2</sup>. b) FE of  $C_2H_4$  on OD-LuO<sub>x</sub> and OD-La-CuO<sub>x</sub> catalysts at 300 mA cm<sup>-2</sup> in 1 M KOH and 1 M KCl; c) FE and partial current densities of  $C_{2+}$  products for OD-La<sub>0.10</sub>-CuO<sub>x</sub> in 1 M KCl at varying current densities; d) FE of CH<sub>4</sub> on OD-CuO<sub>x</sub> and OD-La<sub>0.40</sub>-CuO<sub>x</sub> in 1 M KOH and 1 M KCl at current densities from 100 to 500 mA cm<sup>-2</sup>; e) FE of CH<sub>4</sub> on OD-CuO<sub>x</sub> and various OD-La-CuO<sub>x</sub> catalysts at 400 mA cm<sup>-2</sup> in 1 M KOH and 1 M KCl; f) FE and partial current densities of CH<sub>4</sub> for OD-La<sub>0.40</sub>-CuO<sub>x</sub> in 1 M KOH under different current densities. g) Comparison of the FE<sub>C2+</sub> value of OD-La<sub>0.10</sub>-CuO<sub>x</sub> in this work with other reported high-performance Cu-based electrocatalysts [49–60]. h) Cathodic energy efficiency of C<sub>2</sub>H<sub>4</sub> on OD-La<sub>0.10</sub>-CuO<sub>x</sub> and OD-CuO<sub>x</sub> in 1 M KCl, and i) Cathodic energy efficiency of CH<sub>4</sub> on OD-La<sub>0.40</sub>-CuO<sub>x</sub> in 1 M KCl, and i) Cathodic energy efficiency of CH<sub>4</sub> on OD-La<sub>0.40</sub>-CuO<sub>x</sub> in 1 M KCl, and i) Cathodic energy efficiency of CH<sub>4</sub> on OD-La<sub>0.40</sub>-CuO<sub>x</sub> in 1 M KCl, and i) Cathodic energy efficiency of CH<sub>4</sub> on OD-La<sub>0.40</sub>-CuO<sub>x</sub> in 1 M KCl, and i) Cathodic energy efficiency of CH<sub>4</sub> on OD-La<sub>0.40</sub>-CuO<sub>x</sub> in 1 M KCl, and i) Cathodic energy efficiency of CH<sub>4</sub> on OD-La<sub>0.40</sub>-CuO<sub>x</sub> in 1 M KCl, and i) Cathodic energy efficiency of CH<sub>4</sub> on OD-La<sub>0.40</sub>-CuO<sub>x</sub> in 1 M KCl, in 1 M KOH, at current densities from 100 to 500 mA cm<sup>-2</sup>.

the rich grain boundaries in OD-La<sub>0.10</sub>-CuO<sub>x</sub> (Fig. S6) are likely to enhance C-C coupling by improving \*CO adsorption and H<sub>2</sub>O dissociation, thereby facilitating CO<sub>2</sub>RR to C<sub>2+</sub> products [30]. OD-La<sub>0.10</sub>-CuO<sub>x</sub> also demonstrates superior CO<sub>2</sub>RR-to-C<sub>2+</sub> performance in a neutral electrolyte, reaching C<sub>2+</sub> partial current densities near 350 mA cm<sup>-2</sup> (Fig. 3c).

Beyond achieving excellent  $C_{2+}$  selectivity, the emergence of the La<sub>2</sub>CuO<sub>4</sub> phase and its structure-function relationships with CO<sub>2</sub>RR-to-CH<sub>4</sub> are of great interest. Consequently, online FE tests were conducted at 100–500 mA cm<sup>-2</sup> in both electrolytes to assess the effect of La<sub>2</sub>CuO<sub>4</sub> phase on FE<sub>CH4</sub> (Figs. S18 and S19). Notably, OD-La<sub>0.40</sub>-CuO<sub>x</sub> exhibits no less than 40% FE<sub>CH4</sub> at current densities of 200–500 mA cm<sup>-2</sup> in 1 M KOH (Fig. 3d). However, undoped OD-CuO<sub>x</sub> shows almost no FE<sub>CH4</sub>, achieving less than 1% in both alkaline and neutral electrolytes. FE<sub>CH4</sub> was further examined for CuO<sub>x</sub> and La-CuO<sub>x</sub> catalysts at 400 mA cm<sup>-2</sup> in 1 M KOH. As illustrated in Fig. 3e, catalysts containing La<sub>2</sub>CuO<sub>4</sub> phase (ranging from OD-La<sub>0.13</sub>-CuO<sub>x</sub> to OD-La<sub>0.40</sub>-CuO<sub>x</sub>) significantly promote CO<sub>2</sub> hydrogenation to CH<sub>4</sub>, especially in alkaline environments. OD-La<sub>0.40</sub>-CuO<sub>x</sub> achieves up to 61.5% FE<sub>CH4</sub> at 400 mA cm<sup>-2</sup> in 1 M KOH,

with partial current densities reaching up to 270 mA cm<sup>-2</sup> (Fig. 3f). This can be attributed to the increased La<sub>2</sub>CuO<sub>4</sub> ratio and surface-adsorbed oxygen-containing species, which are considered to provide H<sub>2</sub>O dissociation sites to enhance CO<sub>2</sub>RR-to-CH<sub>4</sub> process [30,61]. However, with further increase in La content, the highest FE<sub>CH4</sub> achieved by OD-La<sub>0.50</sub>-CuO<sub>x</sub> is below 40% in both 1 M KOH and 1 M KCl, accompanied by serious hydrogen evolution reaction (Tables S5 and S6). Besides, comparing alkaline and neutral electrolytes, FE<sub>H2</sub> analysis for OD-La<sub>0.40</sub>-CuO<sub>x</sub> indicates that HER is boosted in neutral 1 M KCl, reducing CH<sub>4</sub> selectivity (Fig. S21).

In summary, the OD-La<sub>0.10</sub>-CuO<sub>x</sub> and OD-La<sub>0.40</sub>-CuO<sub>x</sub> catalysts demonstrate high electron selectivity for producing  $C_2H_4$  and  $CH_4$ , respectively. As shown in Fig. 3g and S22, these electrocatalysts surpass the performance of most reported state-of-the-art Cu-based catalysts for CO<sub>2</sub>RR-to-C<sub>2+</sub> and CO<sub>2</sub>RR-to-CH<sub>4</sub> in a gas-fed flow cell configuration, respectively. The original data supporting these findings are detailed in Tables S7 and S8. Energy efficiency and electrocatalyst stability remain challenges in the CO<sub>2</sub>RR process [62]. Accordingly, the as-prepared electrocatalysts (OD-La<sub>0.10</sub>-CuO<sub>x</sub> and OD-La<sub>0.40</sub>-CuO<sub>x</sub>) were integrated

into a gas-fed flow cell with 1 M KCl and 1 M KOH as electrolytes, and nickel foam as the anode, to evaluate their energy efficiency and stability. As shown in Fig. **3h** and **i**, at 300 and 400 mA cm<sup>-2</sup>, the highest cathodic energy efficiencies for  $C_2H_4$  and  $CH_4$  of 23.0% and 28.2% are

achieved on OD-La<sub>0.10</sub>-CuO<sub>x</sub> and OD-La<sub>0.40</sub>-CuO<sub>x</sub>, respectively, exceeding recent reports of 20.9% for  $C_2H_4$  [63] and 20% for CH<sub>4</sub> [64]. Detailed current density, Faradic efficiency, cathodic energy efficiency, and applied potential post-*iR* compensation for OD-CuO<sub>x</sub> and



**Fig. 4.** Tafel plots of the formation of a)  $C_2H_4$  in 1 M KCl, and b)  $CH_4$  in 1 M KOH for OD-CuO<sub>x</sub>, OD-La<sub>0.10</sub>-CuO<sub>x</sub>, and OD-La<sub>0.40</sub>-CuO<sub>x</sub>. Bode phase plots of c) OD-La<sub>0.10</sub>-CuO<sub>x</sub> in 1 M KCl and d) OD-La<sub>0.40</sub>-CuO<sub>x</sub> in 1 M KOH at the equilibrium potential for CO<sub>2</sub> electrolysis. e) Kinetic isotope effect (KIE) values for CO<sub>2</sub>RR-to-C<sub>2</sub>H<sub>4</sub> on OD-CuO<sub>x</sub> and OD-La<sub>0.40</sub>-CuO<sub>x</sub> measured at 300 mA cm<sup>-2</sup> in 1 M KCl (left) and KIE values for CO<sub>2</sub>RR-to-CH<sub>4</sub> on OD-CuO<sub>x</sub> and OD-La<sub>0.40</sub>-CuO<sub>x</sub> measured at 400 mA cm<sup>-2</sup> in 1 M KOH (right). f) Plots of calculated proton-adsorption pseudo-capacitance (C $\phi$ ) for OD-CuO<sub>x</sub>, OD-La<sub>0.10</sub>-CuO<sub>x</sub>, and OD-La<sub>0.40</sub>-CuO<sub>x</sub> at different potentials in 1 M KOH, with inset showing the equivalent circuit for the single-adsorbate mechanism ( $R_s$ : solution resistance;  $R_{ct}$ : charge transfer resistance;  $R_2$ : hydrogen adsorption resistance; CPE: constant phase angle element).

OD-La<sub>0 10</sub>-CuO<sub>v</sub> in 1 M KCl and OD-CuO<sub>v</sub> and OD-La<sub>0 40</sub>-CuO<sub>v</sub> in 1 M KOH are provided in Tables S9 and S10, respectively. Moreover, OD-La<sub>0.10</sub>-CuO<sub>x</sub> shows stable CO<sub>2</sub>RR-to-C<sub>2</sub>H<sub>4</sub> operation for 16 hours at 300 mA  $\rm cm^{-2}$  in 1 M KCl with a cation exchange membrane. As illustrated in Fig. S23, there is only a slight decrease in  $FE_{C2H4}$  during the stability test, indicating excellent CO2RR stability. The spent OD-La<sub>0.10</sub>-CuO<sub>x</sub> catalyst reaches its highest FE<sub>C2H4</sub> at around 30 minutes and maintains 90% of the highest electron selectivity ( $FE_{C2H4} > 45\%$ ) in subsequent runs in a flow cell. OD-La<sub>0.40</sub>-CuO<sub>x</sub> sustains stable CO<sub>2</sub>RR-to-CH<sub>4</sub> operation for 8 hours at 400 mA  $\text{cm}^{-2}$  in 1 M KOH with an anion exchange membrane. As shown in Fig. S24, despite the hydrophilic nature of OD-La<sub>0.40</sub>-CuO<sub>x</sub> causing slight fluctuations in the applied potential range, it achieves the highest FE<sub>CH4</sub> at around 30 minutes and maintains FE<sub>CH4</sub> over 50% during the first 6 hours. After 8 hours of catalyst stability testing, the gas diffusion electrode (GDE) shows (bi) carbonate precipitation and flooding issues, leading to the blockage of the gas transfer channel in the flow cell, which reduces the electron selectivity for CH<sub>4</sub>.

# 3.3. Kinetic insights into CO<sub>2</sub>RR

Electrocatalytic kinetic analysis is a well-established approach for investigating reaction mechanisms and determining the structurefunction relationships of electrocatalysts. To elucidate the kinetics underlying the enhanced CO<sub>2</sub>RR-to-C<sub>2</sub>H<sub>4</sub>/CH<sub>4</sub> performance of various OD-La-CuO<sub>x</sub> catalysts, Tafel curves for the formation of C<sub>2</sub>H<sub>4</sub> and CH<sub>4</sub> at different partial current densities were plotted. During the CO<sub>2</sub>RR-to-C<sub>2</sub>H<sub>4</sub> process, the Tafel slope for OD-La<sub>0 10</sub>-CuO<sub>x</sub> in 1 M KCl (144 mV  $dec^{-1}$ ) is lower than that of undoped OD-CuO<sub>x</sub> (152 mV dec<sup>-1</sup>) and OD- $La_{0.40}$ -CuO<sub>x</sub> (173 mV dec<sup>-1</sup>), the latter being enriched with  $La_2$ CuO<sub>4</sub> phases due to the increased La doping (Fig. 4a). A similar trend was observed in 1 M KOH, indicating that OD-La<sub>0.10</sub>-CuO<sub>x</sub> has superior initial activity and kinetics for efficient C<sub>2</sub>H<sub>4</sub> production (Fig. S25a). Furthermore, kinetic analysis of the CO<sub>2</sub>RR-to-CH<sub>4</sub> process reveals that the Tafel slope for OD-La $_{0.40}$ -CuOx in 1 M KOH is 85 mV dec $^{-1}$ , lower than that of OD-CuO<sub>x</sub> (88 mV dec<sup>-1</sup>) and OD-La<sub>0.10</sub>-CuO<sub>x</sub> (117 mV  $dec^{-1}$ ) (Fig. 4b). OD-La<sub>0.40</sub>-CuO<sub>x</sub> also exhibits the lowest Tafel slope in 1 M KCl, indicating enhanced electrocatalytic activity, primarily due to accelerated electron transfer from the electrode to catalyst during CO<sub>2</sub> methanation (Fig. S25b).

Moreover, in situ electrochemical impedance spectroscopy (EIS) was employed to assess the electrocatalytic kinetics and ion diffusion capabilities. Figs. 4c, d and S26 show the frequency-dependent changes in the Bode phase plots for OD-La<sub>0.10</sub>-CuO<sub>x</sub> and OD-La<sub>0.40</sub>-CuO<sub>x</sub> under various applied potentials. Generally, the Bode plots show a shift of the phase angle  $(\phi)$  towards higher frequency regions with increasing applied potential, along with a reduction in the peak value. This results in a decrease in Faradaic resistance and an increase in surface reaction rate, aligning with the electrocatalytic processes occurring on the surface [65,66]. The Bode plot reveals a more substantial decline in phase angles for OD-La<sub>0 40</sub>-CuO<sub>x</sub> compared to OD-La<sub>0 10</sub>-CuO<sub>x</sub> in both 1 M KCl (Figs. 4c and S26b) and 1 M KOH (Figs. 4d and S26a) when subjected to applied potentials from -0.36 to -0.86 V. This suggests that the surface of OD-La<sub>0.40</sub>-CuO<sub>x</sub> is more conducive to enhanced charge transfer capability and superior electronic conductivity [67]. As previously analyzed by XPS, this may be attributed to enriched surface oxygen-containing species in OD-La<sub>0.40</sub>-CuO<sub>x</sub> (Fig. 2h).

In this work, KOH and KCl electrolytes were applied in the electrocatalytic CO<sub>2</sub>RR processes, thus anticipating \*H donation primarily from H<sub>2</sub>O dissociation rather than HCO<sub>3</sub> [21]. To understand the role of H<sub>2</sub>O dissociation and \*H transfer in the CO<sub>2</sub>RR-to-C<sub>2+</sub>/CH<sub>4</sub> processes, we examined the kinetic isotope effect (KIE) of hydrogen/deuterium (H/D) on OD-CuO<sub>x</sub>, OD-La<sub>0.140</sub>-CuO<sub>x</sub>, and OD-La<sub>0.40</sub>-CuO<sub>x</sub>. The KIE value for H/D is determined by the ratio of the formation rates of the product before and after substituting H<sub>2</sub>O with D<sub>2</sub>O in the original electrolyte (Eq. S5). Generally, higher KIE values indicates more pronounced effects of  $H_2O$  dissociation and \*H transfer in the rate-determining step (RDS). Typically, KIE values exceeding 1 indicate that  $H_2O$  activation and proton transfer are involved in the RDS [68]. A KIE value greater than 1.5 suggests that \*H attraction significantly affects the reaction rate [69, 70].

To determine the KIE value of  $C_2H_4$ ,  $D_2O$  was used instead of  $H_2O$  in the 1 M KCl electrolyte. As shown in Fig. 4e, the KIE values of  $C_2H_4$  for OD-CuO<sub>x</sub>, OD-La<sub>0.10</sub>-CuO<sub>x</sub>, and OD-La<sub>0.40</sub>-CuO<sub>x</sub> are around 1, specifically 1.13, 1.06 and 0.950, respectively, suggesting there may be no \*Hrelated RDS or multiple parallel \*H-related RDSs in  $C_2H_4$  formation mechanisms [68]. In addition, the KIE value of  $C_2H_4/C_2D_4$  for OD-La<sub>0.40</sub>-CuO<sub>x</sub> is the lowest compared to OD-CuO<sub>x</sub> and OD-La<sub>0.10</sub>--CuO<sub>x</sub>, confirming that the presence of La accelerates H<sub>2</sub>O dissociation to provide \*H for CO<sub>2</sub>RR-to-C<sub>2</sub>H<sub>4</sub>. Similarly, by using D<sub>2</sub>O instead of H<sub>2</sub>O in 1 M KOH, the KIE values of CH<sub>4</sub> on OD-CuO<sub>x</sub>, OD-La<sub>0.10</sub>-CuO<sub>x</sub>, and OD-La<sub>0.40</sub>-CuO<sub>x</sub> are 1.73, 1.66 and 1.48, respectively. This suggests that the CO<sub>2</sub>RR-to-CH<sub>4</sub> kinetics on OD-La<sub>0.40</sub>-CuO<sub>x</sub> is significantly constrained by \*H transfer, as shown in Fig. 4e.

Given the well-established effects of  $H_2O$  dissociation and protonation on the kinetics of  $CO_2RR$ -to- $C_{2+}/CH_4$ , we subsequently investigated the coverage of adsorbed \*H on the catalyst surface using an equivalent circuit for a single-adsorbate mechanism (Armstrong's electric circuit) via *in situ* EIS measurements [71]. The proton-adsorption pseudo-capacitance (C $\varphi$ ) in the second parallel component represents the \*H coverage [11,71,72]. As illustrated in Figs. 4f and S27, C $\varphi$  for OD-La<sub>0.40</sub>-CuO<sub>x</sub> exceeds that of OD-La<sub>0.10</sub>-CuO<sub>x</sub> and OD-CuO<sub>x</sub> across the same applied potential range (-0.4 to -0.9 V vs. RHE) in both 1 M KOH and 1 M KCl. This indicates that OD-La<sub>0.40</sub>-CuO<sub>x</sub>, with its high H<sub>2</sub>O dissociation activity, can provide sufficient \*H for the multistep protonation of CH<sub>4</sub> intermediates. Given that C $\varphi$  increases with rising La content, it is confirmed that doped-La favors proton delivery in CO<sub>2</sub>RR, consistent with the results from the above KIE experiment.

#### 3.4. In situ ATR-SEIRAS observations of CO<sub>2</sub>RR

To explore the mechanisms of dimerization and hydrogenation during CO<sub>2</sub>RR on the as-prepared OD-La-CuO<sub>x</sub> catalysts, time-resolved in situ ATR-SEIRAS was employed to monitor the formation of various intermediates on OD-CuOx control, OD-La<sub>0.10</sub>-CuOx, and OD-La<sub>0.40</sub>- $CuO_x$ , focusing on the pathways leading to  $C_1$  and  $C_{2+}$  products. Fig. 5a illustrates the observed states of surface intermediates on undoped OD-CuO<sub>x</sub> nanoparticles during a continuous 10-minute chronoamperometry test. The absorption peak at  $1260 \text{ cm}^{-1}$ , corresponding to the C=O stretching in the carboxylate radical (\*COOH), intensifies with prolonged electrolysis [73]. The OD-CuO<sub>x</sub> sample exhibits a distinct signal for the CO vibrational frequency ( $\nu$ CO) in \*OCCOH at 1460 cm<sup>-2</sup> indicative of the intermediate in  $C_{2+}$  product formation [18]. Interestingly, a peak at 1556 cm<sup>-1</sup> indicates the presence of \*OCCHO, a key intermediate in C<sub>2</sub>H<sub>4</sub> formation along with \*OCCOH [74]. The IR band at 1640 cm<sup>-1</sup> is commonly associated with the bending mode of the H<sub>2</sub>O molecule (H-O-H) [18]. These observations are consistent with the CO<sub>2</sub>RR product distribution for the OD-CuO<sub>x</sub> control.

Further analysis of the surface intermediate states was conducted using OD-La<sub>0.10</sub>-CuO<sub>x</sub> under identical *in situ* ATR-SEIRAS conditions (Fig. 5b). Continuous chronoamperometry reveals the same distinct absorption peaks near 1260 cm<sup>-1</sup> and 1642 cm<sup>-1</sup>, originating from \*COOH and \*H<sub>2</sub>O, respectively [18,73]. Notably, a broad peak ranging from 1400 to 1470 cm<sup>-1</sup> is primarily composed of two peaks at 1410 cm<sup>-1</sup> and 1445 cm<sup>-1</sup>, which correspond to the symmetric stretching of \*COO<sup>-</sup> and \*C<sub>2</sub>H<sub>4</sub>, respectively [75,76]. The observed \*C<sub>2</sub>H<sub>4</sub> reaction intermediates indicate the formation of C<sub>2</sub>H<sub>4</sub> on OD-La<sub>0.10</sub>-CuO<sub>x</sub>. With continued electrolysis, a peak at 1500 cm<sup>-1</sup> appears, attributed to \*OCCO [18]. Additionally, another time-dependent IR peak is detected at 1550 cm<sup>-1</sup>, which is attributed to adsorbed \*OCCHO [74]. By comparing the intensities of both \*OCCO and \*OCCHO peaks, we find that \*OCCHO dominates the C-C coupling



Fig. 5. Time-dependent electrochemical *in situ* ATR-SEIRAS measurements of a) OD- $CuO_x$ , b) OD- $La_{0.10}$ - $CuO_x$ , and c) OD- $La_{0.40}$ - $CuO_x$  at a potential of -0.8 V vs. RHE. All spectroscopic tests were conducted in a  $CO_2$ -saturated 1 M KCl solution over 10-min  $CO_2$  electrolysis.

process on OD-La<sub>0.10</sub>-CuO<sub>x</sub>. These results confirm that the surface of OD-La<sub>0.10</sub>-CuO<sub>x</sub> is predominantly covered with intermediates crucial for C<sub>2+</sub> product formation, such as \*C<sub>2</sub>H<sub>4</sub>, \*OCCO, and \*OCCHO. The increasing intensity of these peaks with ongoing electrolysis strongly corroborates the excellent performance for CO<sub>2</sub>RR-to-C<sub>2+</sub> discussed earlier.

To further substantiate that varying La-Cu ratios influence product selectivity, *in situ* time-dependent ATR-SEIRAS tests were also

performed on OD-La<sub>0,40</sub>-CuO<sub>x</sub>. A peak associate with \*COOH is found at 1260 cm<sup>-1</sup> [73]. Interestingly, signals indicative of intermediates for C<sub>2+</sub> formation (\*OCCHO, \*OCCO, and \*OCCOH) are weakened or even absent (Fig. 5c). Instead, a peak at 1730 cm<sup>-1</sup> arises from the C=O stretching of \*CHO, a critical intermediate in the CO<sub>2</sub>RR-to-CH<sub>4</sub> pathway [77]. Moreover, a distinct \*CH<sub>2</sub>O signal appears at 1471 cm<sup>-1</sup> from the protonation of \*CHO [78]. Additionally, a peak corresponding to \*CH<sub>3</sub>O is detected at 1400 cm<sup>-1</sup>, produced from further protonation

of \*CH<sub>2</sub>O and serving as a crucial intermediate of electrochemical CO<sub>2</sub> methanation [78]. Throughout constant electrolysis, a prominent broad peak emerges in the IR range of 1600–1700 cm<sup>-1</sup>, particularly around 1650 cm<sup>-1</sup>, related to the H-O-H bending mode of H<sub>2</sub>O molecules [18]. As mentioned above, this broad peak suggests the concurrent activation of H<sub>2</sub>O on the La<sub>0.40</sub>-Cu surface during CO<sub>2</sub>RR. Based on the intermediate information obtained from OD-La<sub>0.40</sub>-CuO<sub>x</sub>, it can be speculated that the adsorbed H<sub>2</sub>O molecules continuously supply protons, enhancing the protonation of intermediates, promoting \*CHO, \*CH<sub>2</sub>O,

and \*CH<sub>3</sub>O formation, and optimizing CO<sub>2</sub>RR kinetics [79]. In summary, the results of *in situ* ATR-SEIRAS depicted in Fig. 5c demonstrate minimal amounts of intermediates for C<sub>2+</sub> product formation on OD-La<sub>0.40</sub>-CuO<sub>x</sub>, but a substantial accumulation of intermediates related to CH<sub>4</sub>, which is consistent with observations in product distribution of electrochemical experiments.



**Fig. 6.** a) Gibbs free energy diagrams for the hydrogenation of \*CHO to \*CH<sub>2</sub>O on OD-CuO<sub>x</sub> surfaces without La doping and on OD-La<sub>0.40</sub>-CuO<sub>x</sub>; b) Gibbs free energy diagrams for the dimerization of \*CHO to \*OCCHO on OD-CuO<sub>x</sub> surfaces without La doping and on OD-La<sub>0.10</sub>-CuO<sub>x</sub>; c&d) Projected density of states (PDOS) of d orbitals associated with \*CH<sub>2</sub>O and \*OCCHO on OD-La<sub>0.40</sub>-CuO<sub>x</sub> and OD-La<sub>0.10</sub>-CuO<sub>x</sub>, respectively, with d-band centers ( $\epsilon_d$ ) indicated by red dashed lines. Adsorption energy diagram of (e) H<sub>2</sub>O dissociation and (f) hydrogen ad-desorption for OD-CuO<sub>x</sub> without La doping, OD-La<sub>0.40</sub>-CuO<sub>x</sub> and OD-La<sub>0.10</sub>-CuO<sub>x</sub>. (g) Proposed reaction mechanisms, illustrating CO<sub>2</sub> dimerization to C<sub>2+</sub> products on OD-La<sub>0.10</sub>-CuO<sub>x</sub> (top) and CO<sub>2</sub> hydrogenation to CH<sub>4</sub> on OD-La<sub>0.40</sub>-CuO<sub>x</sub> (bottom). Atoms are colored as follows: Cu, blue; La, green; O, red; C, brown; and H, pale pink.

# 3.5. DFT calculations of CO<sub>2</sub>RR

To elucidate the reaction mechanisms and product correlations of La-CuO<sub>x</sub> catalysts with varying levels of La doping in CO<sub>2</sub>RR, DFT calculations at the functional/basis level were performed. Based on the structural characteristics observed from the XRD patterns of the La-CuOx catalysts post-reaction (Fig. 2b and c), theoretical models of  $OD-CuO_x$ without La doping, OD-La<sub>0.10</sub>-CuO<sub>x</sub>, and OD-La<sub>0.40</sub>-CuO<sub>x</sub> were constructed (Fig. S28). Two primary reaction pathways for CO<sub>2</sub>RR-to-CH<sub>4</sub> have been documented [80]: namely \*CO  $\rightarrow$  \*CHO  $\rightarrow$  \*CH<sub>2</sub>O  $\rightarrow$  \*CH<sub>3</sub>O  $\rightarrow$  CH<sub>4</sub> and \*CO  $\rightarrow$  \*COH  $\rightarrow$  \*C  $\rightarrow$  \*CH  $\rightarrow$  \*CH<sub>2</sub>  $\rightarrow$  \*CH<sub>3</sub>  $\rightarrow$  CH<sub>4</sub>. To differentiate between these pathways, Gibbs free energy ( $\Delta G$ ) for \*CO  $\rightarrow$ \*CHO and \*CO  $\rightarrow$  \*COH transitions were calculated (Fig. S29). The results indicate that for all three electrocatalysts, the  $\Delta G$  for \*CO  $\rightarrow$  \*CHO is significantly lower than that for  $*CO \rightarrow *COH$ , effectively ruling out the latter pathway as viable for CH<sub>4</sub> formation. Additionally, OD-La<sub>0.10</sub>-CuO<sub>x</sub> displays the lowest  $\Delta G$  for \*CO  $\rightarrow$  \*CHO compared to OD-CuOx without La and OD-La0.40-CuOx, indicating the improved \*CHO formation at moderate La doping levels. The subsequent analysis will prove why OD-La<sub>0.40</sub>-CuO<sub>x</sub> and OD-La<sub>0.10</sub>-CuO<sub>x</sub> exhibit enhanced selectivity towards  $CH_4$  and  $C_{2+}$  products, respectively.

As illustrated in Fig. 6a, the CH<sub>4</sub> formation pathway (\*CHO  $\rightarrow$ \*CH<sub>2</sub>O) on OD-La<sub>0.40</sub>-CuO<sub>x</sub> is energetically more favorable ( $\Delta G$  = -0.50 eV) than on OD-CuO<sub>x</sub> without La doping ( $\Delta G = 0.83 \text{ eV}$ ) and OD-La<sub>0.10</sub>-CuO<sub>x</sub> ( $\Delta G = 1.48$  eV). This suggests that intermediates conducive to CH<sub>4</sub> formation are more readily generated on OD-La<sub>0.40</sub>-CuO<sub>x</sub>. Furthermore, when comparing the reaction energy barriers for the CO<sub>2</sub>RR-to-C<sub>2+</sub> process among OD-La<sub>0.10</sub>-CuO<sub>x</sub>, OD-La<sub>0.40</sub>-CuO<sub>x</sub>, and OD- $CuO_x$  (Fig. 6b), it is evident that the energy barrier for the \*CHO  $\rightarrow$ \*OCCHO process on OD-La<sub>0.10</sub>-CuO<sub>x</sub> (1.10 eV) is lower than that of OD- $CuO_x$  (1.35 eV) and OD-La<sub>0.40</sub>-CuO<sub>x</sub> (1.47 eV). This indicates that OD-La<sub>0.10</sub>-CuO<sub>x</sub> has optimal adsorption of \*OCCHO. Three widely accepted C-C coupling pathways include CO-CO, CO-CHO and CO-COH [81]. Fig. S29 suggests that the CO-COH pathway is unlikely to occur due to a high energy barrier for \*CO hydrogenation to form \*COH. Additional comparisons of computed reaction energies between CO-CO and CO-CHO pathways (Fig. S30) indicate that the \*OCCHO formation pathway is spontaneous for both OD-CuOx without La doping and OD-La<sub>0.10</sub>-CuO<sub>x</sub> (-1.46 eV and -1.82 eV), with substantially lower energy barriers than those for the \*OCCO formation pathway (1.76 eV and -0.60 eV), underscoring the dominance of the \*CHO  $\rightarrow$  \*OCCHO pathway among the three C-C coupling routes. An examination of Gibbs free energy diagrams for CO2RR-to-CH4 and C2+ products reveals a close correlation between CO2RR products on different OD-La-CuOx catalysts and the divergence of \*CHO and the formation of critical intermediates (i.e., \*CH<sub>2</sub>O and \*OCCHO), which agrees with in situ ATR-SEIRAS observations (Fig. 5).

To understand the interactions between catalyst surfaces and key intermediates (\*OCCHO/ \*CH<sub>2</sub>O) involved in forming  $C_{2+}$  products and CH<sub>4</sub>, an analysis of the projected density of states (PDOS) was conducted. The PDOS illustrated in Fig. 6c and d reveal significant hybridization between the OD-La<sub>0.40</sub>-CuO<sub>x</sub> (3d) and \*CH<sub>2</sub>O (2p) orbitals, as well as between OD-La<sub>0.10</sub>-CuO<sub>x</sub> (3d) and \*OCCHO (2p) orbitals, respectively. This implies robust binding interactions between the catalyst surfaces and these intermediates. Furthermore, the d-band centers ( $\varepsilon_d$ ) of both OD-La<sub>0.40</sub>-CuO<sub>x</sub> and OD-La<sub>0.10</sub>-CuO<sub>x</sub> are situated further from the Fermi energy level ( $E_F = 0$  eV) and are more negative compared to the CuO<sub>x</sub> without La doping. This observation indicates that \*OCCHO and \*CH<sub>2</sub>O exhibit weaker binding affinities and hence less resistance to desorption on OD-La<sub>0.40</sub>-CuO<sub>x</sub> and OD-La<sub>0.10</sub>-CuO<sub>x</sub>, thereby lowering the reaction barriers for CO<sub>2</sub> electroreduction to C<sub>2+</sub> product and CH<sub>4</sub>.

Inspired by the results of KIE and *in situ* EIS, we explored how adsorbed hydrogen (\*H) affects the \*CHO divergence. The concentration of \*H on La-CuO<sub>x</sub> catalysts featuring different hybridized hetero-structures was determined by calculating the free energy changes

associated with water dissociation ( $\Delta G(H_2O)$ ) and hydrogen addesorption ( $\Delta G(H)$ ) on OD-CuO<sub>x</sub> without La doping, OD-La<sub>0.10</sub>-CuO<sub>x</sub> and OD-La<sub>0.40</sub>-CuO<sub>x</sub> (Figs. S31–33). The  $\Delta G(H_2O)$  value of OD-La<sub>0.40</sub>-CuO<sub>x</sub> is 1.19 eV, which is lower than that of OD-La<sub>0.10</sub>-CuO<sub>x</sub> (1.72 eV) and OD-CuO<sub>x</sub> without La doping (2.79 eV) (Fig. 6e). This suggests that increased La doping enhances the thermodynamic favorability of H<sub>2</sub>O dissociation in La-CuO<sub>x</sub> catalysts, resulting in higher \*H availability for the electrocatalytic process. In contrast, the absolute value of  $\Delta G(H)$  for OD-La<sub>0.40</sub>-CuO<sub>x</sub>, at 0.14 eV, is substantially lower than that of OD-CuO<sub>x</sub> without La doping and OD-La<sub>0.10</sub>-CuO<sub>x</sub> (0.74 eV and 0.82 eV, respectively) (Fig. 6f). This reveals the high activity of OD-La<sub>0.40</sub>-CuO<sub>x</sub> for HER, aligning with electrochemical test results (Fig. S21).

Therefore, based on insights from in situ spectroscopy and theoretical analyses, it can be concluded that OD-La<sub>0.10</sub>-CuO<sub>x</sub> and OD-La<sub>0.40</sub>-CuO<sub>x</sub> exhibit distinctive capacities for water dissociation, resulting in varied \*H availability on their respective surfaces. As illustrated in Fig. 6g, the presence of moderate \*H on the OD-La<sub>0.10</sub>-CuOx surface and abundant \*H on the OD-La<sub>0.40</sub>-CuO<sub>x</sub> surface lead to different \*CHO conversion pathways, lowering energy barriers for both dimerization (\*CHO  $\rightarrow$ \*OCCHO) and hydrogenation (\*CHO  $\rightarrow$  \*CH<sub>2</sub>O) processes, thereby achieving enhanced CO<sub>2</sub>RR performance towards C<sub>2+</sub> products and CH<sub>4</sub>. These differences can be attributed to the varied dependence on proton transfer in the RDS for CO<sub>2</sub>RR-to-C<sub>2+</sub>/CH<sub>4</sub>. According to KIE results, the CO<sub>2</sub>RR-to-C<sub>2+</sub> process may lack a distinct proton-related RDS or involve multiple parallel proton-related RDSs, whereas the CO2RR-to-CH4 process is significantly influenced by proton transfer in its RDS. These findings are in line with observations from electrochemical performance tests and kinetic studies.

#### 4. Conclusions

In summary, we have developed a series of OD-La-CuOx catalysts with tailored adsorbed hydrogen (\*H) to achieve controlled industrialcurrent-density CO2 conversion to C2+ products and CH4. The OD-La<sub>0.10</sub>-CuO<sub>x</sub> catalyst exhibits outstanding performance of CO<sub>2</sub>RR to C<sub>2+</sub> products in 1 M KCl, reaching a Faradaic efficiency (FE) of 80.4% for  $C_{2+}$  products at a current density of 300 mA cm<sup>-2</sup>. Conversely, with increased La doping in OD-La<sub>0.40</sub>-CuO<sub>x</sub>, a maximum FE of 61.5% for CH<sub>4</sub> is achieved at 400 mA  $\rm cm^{-2}$  in 1 M KOH. Both experimental and kinetic studies reveal that increased La doping alters the phase composition of OD-La-CuO<sub>x</sub> catalysts and correlates positively with higher concentrations of \*H, allowing for tunable electron selectivity and kinetics of CO2RR products. Notably, in situ ATR-SEIRAS and DFT studies demonstrate that the OD-La-CuOx catalysts possess varying H<sub>2</sub>O dissociation capacities, accounting for their variations in \*H. Additionally, this modulation strategy can guide the reaction pathway toward either \*CHO dimerization or hydrogenation, thereby lowering reaction barriers for producing target C<sub>2+</sub> products and CH<sub>4</sub>. OD-La<sub>0.10</sub>-CuO<sub>x</sub>, with its moderate H<sub>2</sub>O dissociation capacity, favors the \*CO  $\rightarrow$  \*CHO and \*CHO  $\rightarrow$  \*OCCHO pathways, enhancing C-C coupling and leading to C<sub>2+</sub> production. In contrast, OD-La<sub>0.40</sub>-CuO<sub>x</sub>, characterized by the lowest H<sub>2</sub>O dissociation energy barrier, enhances proton transfer, promoting  $CO_2RR$ -to- $CH_4$  via the \* $CHO \rightarrow *CH_2O$  pathway. The findings from this study provide a promising strategy for optimizing the selectivity of target CO<sub>2</sub>RR products at industrial current densities.

#### CRediT authorship contribution statement

Zeyu Guo: Conceptualization, Methodology, Investigation, Formal analysis, Writing – original draft. Huiwen Zhu: Data curation, Validation, Methodology. Zijun Yan: Formal analysis, Data curation. Lei Lei: Methodology, Formal analysis. Degao Wang: Resources, Writing – review & editing. Ziyun Xi: Investigation, Data curation. Yirui Lian: Formal analysis, Data curation. Jiahui Yu: Data curation, Methodology, Funding acquisition. Kam Loon Fow: Supervision, Writing – review & editing. Hainam Do: Supervision, Writing – review & editing. Jonathan **D. Hirst:** Supervision, Writing – review & editing. **Tao Wu:** Supervision, Resources, Writing – review & editing. **Mengxia Xu:** Funding acquisition, Supervision, Writing – review & editing.

#### **Declaration of Competing Interest**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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#### Declaration of competing interest

The authors declare no competing financial interest.

## Appendix. Supporting Information

Supplementary data associated with this article can be found in the attached file.

#### Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at doi:10.1016/j.apcatb.2024.124839.

#### **Data Availability**

Data will be made available on request. No data was used for the research described in the article.

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